

Observations of creep of polar firn at different temperatures

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Abstract

To improve our understanding of firn compaction and deformation processes, constant-load compressive creep tests were performed on specimens from a Summit, Greenland (72°35' N, 38°25' W) firn core that was extracted in June, 2017. Cylindrical specimens were tested at temperatures of -5°C , -18°C and -30°C from depths of 20 m, 40 m and 60 m at stresses of 0.21 MPa, 0.32 MPa and 0.43MPa, respectively. The microstructures were characterized before and after creep using both X-ray micro-computed tomography (micro-CT) and thin sections viewed between optical crossed polarizers. The results of these experiments comprise a novel data set on the creep of firn at three depths of a firn column at three different temperatures, providing useful calibration data for firn model development. Examining the resulting strain vs. time and strain vs. strain rate curves from the creep tests revealed the following notable features. First, the time exponent k was found to be 0.34–0.69 during transient creep, which is greater than the 0.33 usually observed in fully-dense ice. Second, the strain rate minimum (SRmin) in secondary creep occurred at a greater strain from specimens with lower density and at higher temperatures. Third, tertiary creep occurred more easily for the lower-density specimens at greater effective stresses and higher temperatures, where strain softening is primarily due to recrystallization. Fourth, the SRmin is a function of the temperature for a given firn density. Lastly, we developed empirical equations for inferring the SRmin, as it is difficult to measure during creep at low temperatures. The creep behaviors of polar firn, being essentially different from full-density ice, imply that firn densification is an indispensable process within the snow-to-ice transition, particularly firn deformation at different temperatures connected to a changing climate.

32 **Keywords:** Firn densification; Creep; Activation energy; Cryospheric micro-CT; Temperature

1. Introduction

Understanding firn compaction and densification experimentally is critical for developing physics-based firn models that are necessary for many glaciological applications. For example, such models are essential for reconstructing ice-core paleoclimate records by simulating the lock-in depth of gases and the smoothing of climate signals (Schwander et al., 1997; Goujon et al., 2003). They are also crucial for interpreting ice-mass changes from satellite altimetry data, as they allow for the accurate correction of firn air content and surface elevation changes not related to underlying ice dynamics (Ligtenberg et al., 2011; Simonsen et al., 2013). However, the firn models used for these applications are empirical and are known to perform poorly outside of their calibration range (Lundin et al., 2017). Thus, a better understanding of firn compaction is necessary to refine firn models for these important glaciological applications. Laboratory compressive experiments on firn and ice improve our understanding of their respective flow laws and advance the development of firn models under a range of conditions. The rheology of polycrystalline ice, particularly its temperature-dependent creep deformation, is a cornerstone of glaciological modeling. Numerous studies have established a robust framework for understanding ice deformation, primarily through laboratory creep experiments (e.g. Glen, 1955; Weertman, 1983; Budd and Jacka, 1989; Durham et al., 2001; Goldsby and Kohlstedt, 2001; Petrenko and Whitworth, 1999). This body of work has confirmed that ice creep is strongly governed by temperature, typically described by an Arrhenius relationship with a well-constrained activation energy for grain-scale processes like dislocation glide and climb (e.g. Jacka, 1984; Hooke, 2005). In contrast, the mechanical behavior of firn, the intermediate porous material between snow and glacial ice, remains comparatively poorly characterized, especially with respect to temperature.

55 The experimental observations are interpreted by drawing parallels between firn deformation and
56 the mechanical properties of its constituent material, polycrystalline ice. This connection is
57 formalized through a poromechanics approach, where the behavior of the porous firn is derived
58 from that of the ice skeleton using continuum mechanics and homogenization principles
59 (Scapozza and Bartelt, 2003; Gagliardini and Meyssonier, 2000; Coussy, 2004; Hutter and Johnk,
60 2004; Srivastava et al., 2010; Theile et al., 2011). While numerous studies have investigated ice
61 deformation (e.g. Steinemann, 1954; Maeno and Ebinuma, 1983; Li et al., 1996; Jacka and Li,
62 2000; Song et al., 2006a, 2006b, 2008; Treverrow et al., 2012; Hammonds and Baker, 2016, 2018)
63 and firn deformation (e.g. Landauer, 1958; Mellor, 1975; Salm, 1982; Ambach and Eisner, 1985;
64 Meussen et al., 1999; Bartelt and von Moos, 2000; Theile et al., 2011; Li and Baker, 2021, 2022a),
65 existing firn data are sparse and fragmented. A critical knowledge gap persists in the systematic
66 experimental quantification of firn's mechanical response across a broad range of temperatures.
67 Temperature is a first-order control on firn densification and deformation rates, yet most
68 laboratory studies have been conducted at a limited number of isothermal conditions, often
69 focused on a single density or at temperatures near the melting point (e.g. Mellor, 1975; Maeno
70 and Ebinuma, 1983). Consequently, there is a pronounced lack of experimental data necessary to
71 derive the systematic activation energy for the creep of firn over its full density spectrum. This
72 parameter is not merely a scalar but is likely a function of density, microstructure, and the
73 dominant deformation mechanism (compaction versus shear), transitioning from grain-boundary
74 sliding in low-density firn to dislocation creep in high-density firn and ice (Hammonds and Baker,
75 2018; Li, 2022; Li and Baker, 2022a). The absence of comprehensive, temperature-variable creep
76 data for firn across its density range renders it insufficient for constraining the

77 temperature-dependence terms in modern, physics-based firn models. Our work fills this gap via
78 X-ray micro-computed tomography-analyzed mechanical examinations, e.g. a systematic series of
79 constant-stress creep experiments on firn cores of varying density, conducted across a thermally
80 controlled range from -30°C to -5°C . This allows for the direct determination whether the
81 apparent activation energy is a function of density, thereby providing the essential experimental
82 foundation needed to improve predictions of firn densification in ice-sheet and glacier models.
83 Notably, the mechanical behavior of two-phase flow coupling the airflow with the ice matrix
84 deformation has not yet been performed experimentally hitherto, even though the role of the
85 microstructures of firn on airflow has been studied (Albert et al., 2000; Courville et al., 2010;
86 Adolph and Albert, 2014). This difficulty is largely due to the limitations of the observation
87 techniques of nondestructive visualization of the microstructures during snow and firn
88 deformation. Thus, caution should be taken when extending the conclusions to ice sheet and
89 glacier scales from sample laboratory experiments. Macroscopically, the creep of firn obeys a
90 power-law dependence of the strain rate on the stress at constant stresses and temperature, similar
91 to that of full-density ice (Li and Baker, 2022a). Note that both the diffusivity and permeability of
92 the air in the pores (Albert et al., 2000; Courville et al., 2010; Adolph and Albert, 2014) impact
93 heat conduction of the ice matrix, and hence the grain growth. This is tightly tied to the
94 micro-mechanisms, e.g. grain-boundary and lattice diffusion of the ice crystals (Li and Baker,
95 2021), superplastic deformation and inter-particle sliding from dislocation motion in the ice necks
96 (Bartelt and Von Moos, 2000), and likely rearrangement of the ice particles (Perutz and Seligman,
97 1939; Anderson and Benson, 1963; Ebinuma and Maeno, 1987).

98

99 Through experiments on isotropic ice samples subjected to uni-axial compaction at octahedral
100 stresses of 0.1–0.8 MPa and temperatures from -45°C to -5°C , Jacka and Li (2000) determined
101 the mechanisms involved in the empirical *power-law flow*, which was derived by Glen (1955) for
102 stresses ranging from 0.1–1 MPa at temperatures spanning from -13°C to the melting-point. They
103 found that dynamic recrystallization predominated at higher temperatures and stresses, whereas
104 crystal rotation governed at lower temperatures and stresses. Later, Goldsby and Kohstedt (2001)
105 found that ice could exhibit *superplastic flow*, which depends inversely on the grain size,
106 particularly for fine-grained ice, while both dislocation creep and basal slip-limited creep were
107 unrelated to the grain size at stresses of 0.1 MPa or less over a wide range of temperatures.
108 Moreover, Baker and Gerberich (1979) reported that the apparent activation energy for creep for
109 polycrystalline ice, which was derived from tests at constant stress and temperatures ranging from
110 -40°C to -5°C , increased with increasing volume fraction of inclusions (bubbles, impurities, dust,
111 and air clathrate hydrates). Such inclusions governed the evolution of grain size related to thermal
112 activations. The activation energies for the creep of snow and ice have been determined by a
113 number of authors, and values ranging from 58.6–113 kJ mol^{-1} were obtained under both uniaxial
114 and hydrostatic experiments for snow with a density of $\sim 400 \text{ kg m}^{-3}$ at -13.6°C to -3.6°C
115 (Landauer, 1958); 44.8–74.5 kJ mol^{-1} from snow with densities of 440–830 kg m^{-3} at -34.5°C to $-$
116 0.5°C (Mellor and Smith, 1966); $\sim 72.9 \text{ kJ mol}^{-1}$ for firn with a density of 320–650 kg m^{-3} at the
117 South Pole (Gow, 1969); $69 \pm 5 \text{ kJ mol}^{-1}$ for a mean snow density of $423 \pm 8 \text{ kg m}^{-3}$ at -19°C to
118 -11°C (Scapozza and Bartelt, 2003); the 78 kJ mol^{-1} from polycrystalline ice compression
119 deformation at a temperature of -10°C (Duval and Ashby, 1983); $\sim 60 \text{ kJ mol}^{-1}$ for artificial and
120 natural ice at the South Pole (Pimienta and Duval, 1987); and $78 \pm 4 \text{ kJ mol}^{-1}$ for monocrystal ice

121 at -20°C to -4.5°C and $75 \pm 2 \text{ kJ mol}^{-1}$ for bicrystal ice at -15°C to -4.5°C (Homer and Glen,
122 1978). In summary, the flow law of polycrystalline ice and firn depends on the effects of
123 recrystallization, grain size, inclusions (Mellor and Testa, 1969; Vickers and Greenfield, 1968;
124 Barnes et al., 1971; Baker and Gerberich, 1979; Goodman et al., 1981), and the temperature.

125
126 With advanced observation techniques, the relevant microstructural parameters of snow and firn
127 have been characterized by a number of scientists (Arnaud et al., 1998; Coleou et al., 2001; Flin et
128 al., 2004; Wang and Baker, 2013; Wiese and Schneebeli, 2017; Li, 2022). Using X-ray
129 micro-computed tomography (micro-CT), Li and Baker (2022b) characterized metamorphism
130 from snow to depth hoar under opposing temperature gradients. Only rarely has work been
131 performed on the co-effects of temperature and stress on the densification of firn while
132 simultaneously visualizing the microstructural changes using a micro-CT. For example, Schleef et
133 al. (2014) reported that densification under varying conditions of overburden stress and
134 temperature from natural and laboratory-grown new snow showed a linear relationship between
135 density and the specific surface area (SSA). To this end, the aim of our present work is to
136 investigate the temperature dependence of the creep of polar firn and relate this to the change of
137 microstructure determined using micro-CT studies on firn obtained from Summit, Greenland in
138 2017. As is well known, temperature is a key parameter affecting the flow of firn and ice, and
139 plays a determined role in their deformation, especially for polythermal and temperate glaciers.
140 Due to the great difficulty of analyzing firn and ice deformation with the presence of liquid water,
141 this work focuses on the firn creep from the dry snow zone, i.e., areas without meltwater, at
142 different temperatures.

143

144 2. Samples and measurements

145 2.1 *Samples*

146 Three cylindrical samples (22 ± 0.5 mm diameter; 50 ± 0.5 mm high) were produced at each of
147 three depths of 20 m, 40 m and 60 m from the same 2017 Summit, Greenland firn core that was
148 studied in Li and Baker (2022a). Both the densities and porosities of these above samples are
149 typical of values in the snow-to-ice transition zone as introduced in Section 1. It is important to
150 note that the reduction in effective stress with increasing depth is evident in samples taken from
151 these three specified depths (**Appendix A**). Before creep testing, one cylindrical firn samples
152 from each depth was stored at a temperature of $-5 \pm 0.5^\circ\text{C}$, $-18 \pm 0.5^\circ\text{C}$, and $-30 \pm 0.5^\circ\text{C}$ for two
153 days to achieve thermal equilibrium (Li and Baker, 2022a). It's also important to note that firn is a
154 heterogeneous material that can have variations in layering, fabric, grain size, and impurity
155 concentration across short distances. Thus, care was taken to extract the three replicate samples
156 from the core at each depth as closely as possible to reduce the variability in their initial
157 conditions.

158

159 2.2 *Creep measurements*

160 Three home-built creep jigs were placed in individual Styrofoam boxes in three different cold
161 rooms that were held at temperatures of $-5 \pm 0.5^\circ\text{C}$, $-18 \pm 0.5^\circ\text{C}$ and $-30 \pm 0.5^\circ\text{C}$. Each creep jig
162 consists of an aluminum base plate and three polished aluminum-guide rails passing through
163 linear bearings that hold the upper aluminum loading plate (**Figure 1**). A linear voltage
164 differential transducer (LVDT-Omega LD-320: resolution of 0.025%; linearity error of $\pm 0.15\%$

of full-scale output), parallel to the three aluminum-guide rails, was located adjacent to the center of the upper plate, and fixed firmly using a screw through the plate (**Figure 1**) for measuring the displacement during a test. The displacement was logged every 5 seconds using a Grant SQ2010 datalogger (accuracy of 0.1%). Temperatures were logged at 300-second time intervals over the entire test period, using a k-type thermocouple (Omega RDXL4SD thermistor: resolution of 0.1°C) that was mounted inside each box. In this work, specimens were tested at temperatures of $-5 \pm 0.2^\circ\text{C}$, $-18 \pm 0.2^\circ\text{C}$ and $-30 \pm 0.2^\circ\text{C}$ from depths of 20 m (applied stress 0.21 MPa), 40 m (0.32 MPa) and 60 m (0.43 MPa). There are smaller error bars for the temperature of the specimens than the room temperature because the creep jigs were in insulated Styrofoam boxes. The stresses were chosen based on experience from previous tests (Li and Baker, 2022a) in order to give measurable creep rates in a reasonable time.

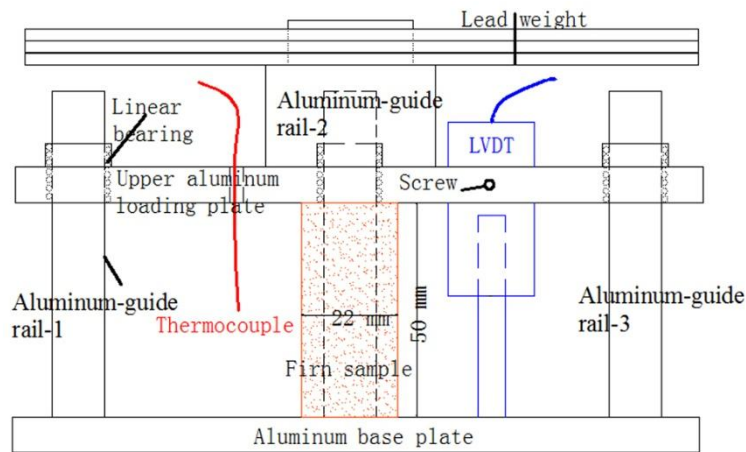


Figure 1: Schematic illustrating the home-built compressive creep jigs. More details can be found in (Li & Baker, 2022a).

2.3 X-ray micro-computed tomography (micro-CT)

Each specimen at each depth and temperature combination was scanned using a Skyscan 1172 micro-CT, before and after creep testing. Each micro-CT scan lasted ~2 h. The cubic Volume of Interest (VOI, a side length of 8 mm) was taken from near the center of the firn specimen as conducted in Li and Baker (2022a). The microstructural parameters obtained from the micro-CT data are the SSA, the mean structure thickness of the ice matrix (S.Th), the area-equivalent circle diameter of the pores (ECDa), the total porosity (TP), the closed porosity (CP), and the structure model index (SMI). The SSA (mm^{-1}) is the ratio of the ice surface area to total firn volume (ice plus air) in a VOI analytical element, and is calculated using the hexahedral marching cubes algorithm via CTAn software (Wang and Baker, 2013). It characterizes the thickness and complexity of the firn microstructure. Changes in SSA indicate a change in free energy of the ice surfaces, the decrease of which represents the occurrence of sintering-pressure. The S.Th (mm) is the mean structure thickness of an ice matrix (Hildebrand and Ruegsegger, 1997), which represents the characteristic size of an ice particle in the firn, where the ice particle consists of one or many crystals or grains. It is measured based on the largest sphere diameter that encloses a point in the ice matrix and is completely bounded within solid surfaces. The ECDa (mm) is the diameter of a circle having the same area as the average for all pores in the VOI from the 2-D binary images, indicative of the characteristic size for the void space (Adolph and Albert, 2014). The TP (%) is the ratio of the pore volume, including both open and closed pores, to the total VOI. The CP (%) is the ratio of the volume of the closed pores to the total volume of solid plus closed pores volume in a VOI, while the open porosity (%) is the ratio of the volume of the open pores to the total VOI. The SMI is calculated based on the dilation of a 3-D voxel model (Hildebrand and

204 Ruegsegger, 1997) $SMI = 6(S' \times V)/S^2$, where S' is the change in the surface area due to
205 dilation, and V and S are the object volume and surface area, respectively. It indicates the
206 prevalent ice curvature, negative values of which represent a concave surface, e.g. the hollow air
207 structure surrounded by an ice matrix. The more negative the SMI value, the more spherical the
208 pore. Notably, the micro-CT-derived density of each specimen agrees well with the bulk density
209 measured using the mass-volume approach (Li and Baker, 2021).

210

211 *2.4 Thin section preparation and imaging*

212 Thin sections for optical photographs before and after creep testing were cut from bulk specimens,
213 one side of which was first smoothed with a microtome. This side was then frozen onto a glass
214 plate ($100 \times 60 \times 2$ mm) by dropping supercooled gas-free water along its edges. Its thickness
215 was reduced to ~ 2 mm by a band saw, and finally thinned further to a uniform thickness of ~ 0.5
216 mm using a microtome. Images were captured using a digital camera after each thin section was
217 placed on a light table between a pair of crossed polarizing sheets.

218

219 **3 Results and discussion**

220 *3.1 Microstructures before creep*

221 Increasing firm density with increasing depth from either of the -5°C , -18°C , and -30°C
222 specimens can be readily recognized by visual inspection of the micro-CT 3-D reconstructions of
223 the firm microstructure (**Figure 2**). Correspondingly, the microstructural parameters, with the
224 exception of the CP, changed monotonically with increasing depth at each temperature, e.g. the $-$
225 30°C samples increased in density from $591 \pm 1.4 \text{ kg m}^{-3}$, to $683 \pm 4.2 \text{ kg m}^{-3}$, to $782 \pm 1.5 \text{ kg m}^{-3}$,

226 decreased in SSA from $4.64 \pm 0.04 \text{ mm}^{-1}$, to $3.3 \pm 0.06 \text{ mm}^{-1}$, to $2.39 \pm 0.01 \text{ mm}^{-1}$, and decreased
227 in TP from $35.6 \pm 0.05\%$, $25.6 \pm 0.4\%$, to $14.8 \pm 0.2\%$ at 20, 40, and 60 m, respectively (**Table 1**).
228 These above changes are similar to those previously observed in this firn core (Li and Baker,
229 2022a), implying that the sintering-pressure mechanism plays a crucial role in the densification of
230 polar firn due to the increasing overburden of snow and firn with increasing depth. However, the
231 microstructures of the samples from the three temperatures at each depth show little variability
232 and do not monotonically change with temperature, e.g. at 20 m depth the -5°C , -18°C , and $-$
233 30°C samples having densities of $589 \pm 1.3 \text{ kg m}^{-3}$, $615 \pm 2.5 \text{ kg m}^{-3}$, and $591 \pm 1.4 \text{ kg m}^{-3}$, and

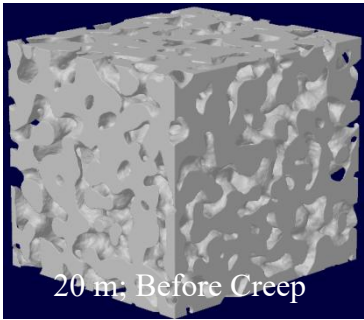
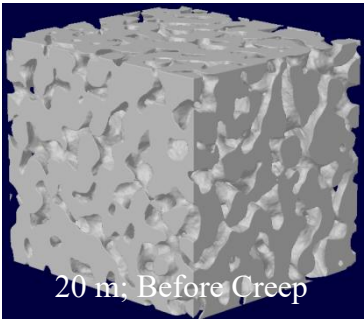
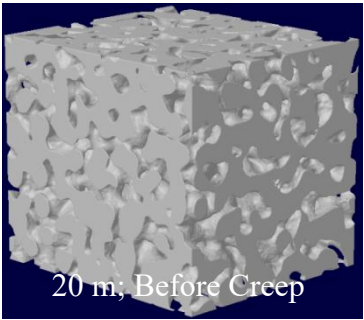
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-5°C

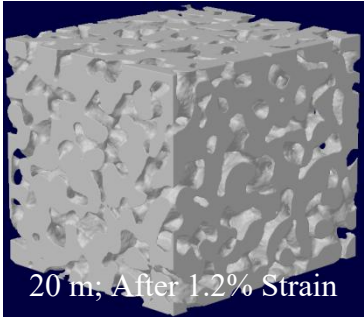
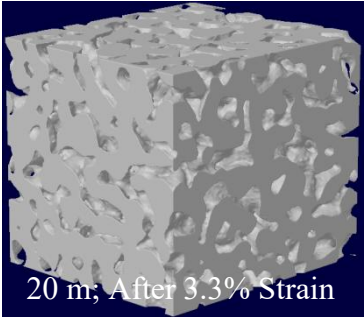
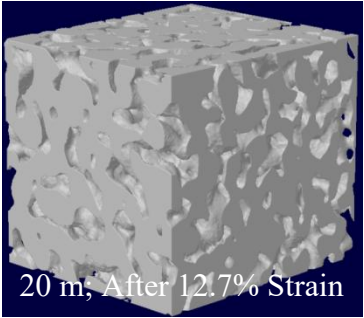
-18°C

-30°C

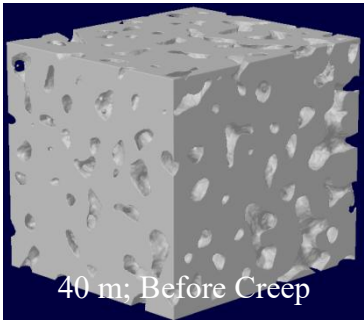
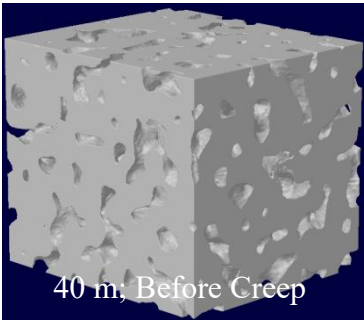
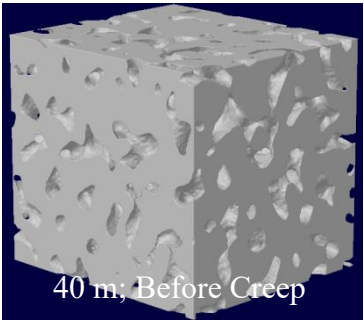
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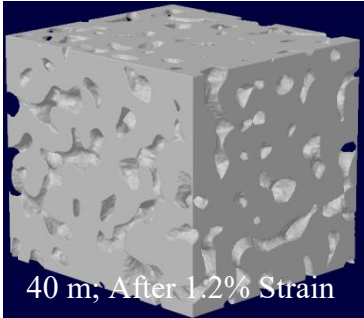
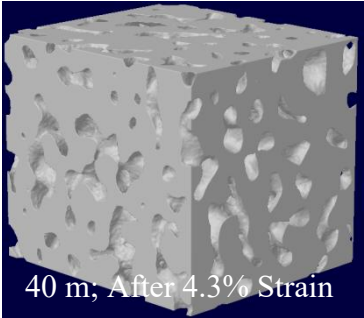
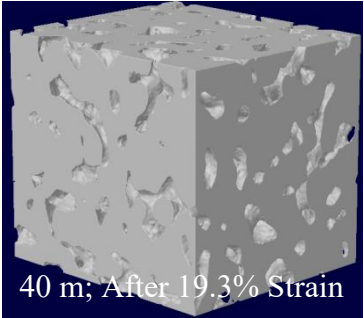
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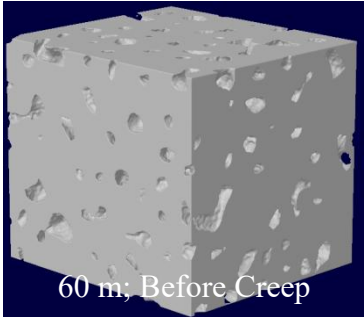
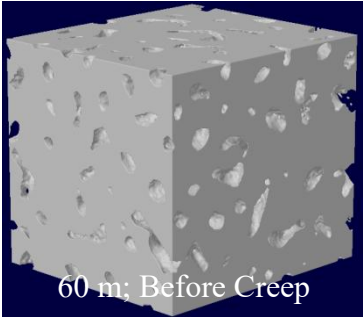
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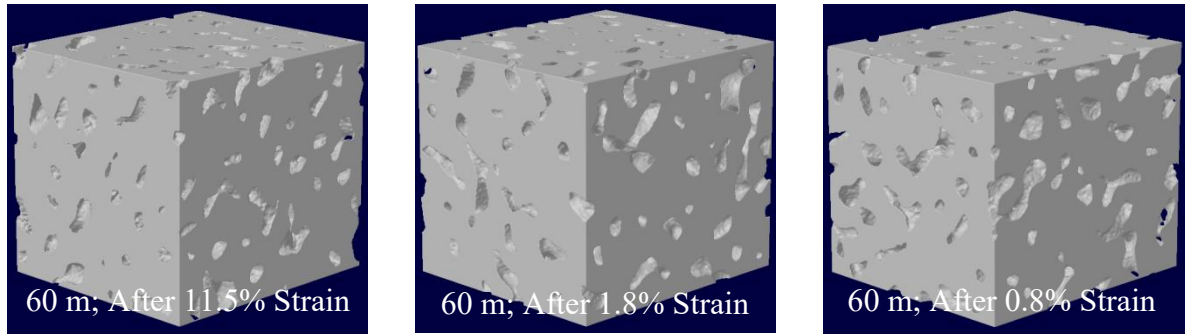


Figure 2: Micro-CT 3-D reconstructions (the side length of each cubic volume of interest is 8 mm) of specimens before and after creep testing at the depths and temperatures shown. Grey voxels represent ice in the firm structure.

Table 1. Microstructural parameters derived from Micro-CT for samples at -5°C , -18°C , and -30°C from depths of 20 m, 40 m, and 60 m before creep.

20m							
T $^{\circ}\text{C}$	Density kg m^{-3}	SSA mm^{-1}	S.Th mm	TP %	CP %	SMI	ECDa mm
-5	589 ± 1.3	4.74 ± 0.03	0.732 ± 0.001	35.9 ± 0.08	0.03 ± 0.002	-0.31 ± 0.04	1.07 ± 0.005
-18	615 ± 2.5	4.51 ± 0.04	0.758 ± 0.001	33.1 ± 0.2	0.01 ± 0.001	-0.57 ± 0.01	0.995 ± 0.013
-30	591 ± 1.4	4.64 ± 0.04	0.747 ± 0.004	35.6 ± 0.05	0.02 ± 0.001	-0.27 ± 0.05	1.09 ± 0.004
40m							
T $^{\circ}\text{C}$	Density kg m^{-3}	SSA mm^{-1}	S.Th mm	TP %	CP %	SMI	ECDa mm
-5	685 ± 1.4	3.26 ± 0.04	0.95 ± 0.004	25.5 ± 0.1	0.015 ± 0.001	-1.85 ± 0.11	0.857 ± 0.005
-18	669 ± 0.6	3.41 ± 0.04	0.914 ± 0.003	27.1 ± 0.03	0.01 ± 0.001	-1.69 ± 0.09	0.892 ± 0.002
-30	683 ± 4.2	3.30 ± 0.06	0.94 ± 0.01	25.6 ± 0.4	0.021 ± 0.005	-1.87 ± 0.13	0.86 ± 0.01
60m							
T $^{\circ}\text{C}$	Density kg m^{-3}	SSA mm^{-1}	S.Th mm	TP %	CP %	SMI	ECDa mm
-5	790 ± 1.0	2.34 ± 0.03	1.1 ± 0.003	14.0 ± 0.1	0.11 ± 0.01	-4.81 ± 0.22	0.594 ± 0.003
-18	780 ± 0.5	2.37 ± 0.03	1.08 ± 0.01	15.2 ± 0.06	0.014 ± 0.007	-4.8 ± 0.2	0.632 ± 0.001
-30	782 ± 1.5	2.39 ± 0.01	1.076 ± 0.0003	14.8 ± 0.2	0.02 ± 0.001	-4.92 ± 0.14	0.639 ± 0.002

Note: SSA is the specific surface area, S.Th is the structure thickness, TP is the total porosity, CP

is the closed porosity, SMI is the structure model index, and ECDA is the area-equivalent circle diameter.

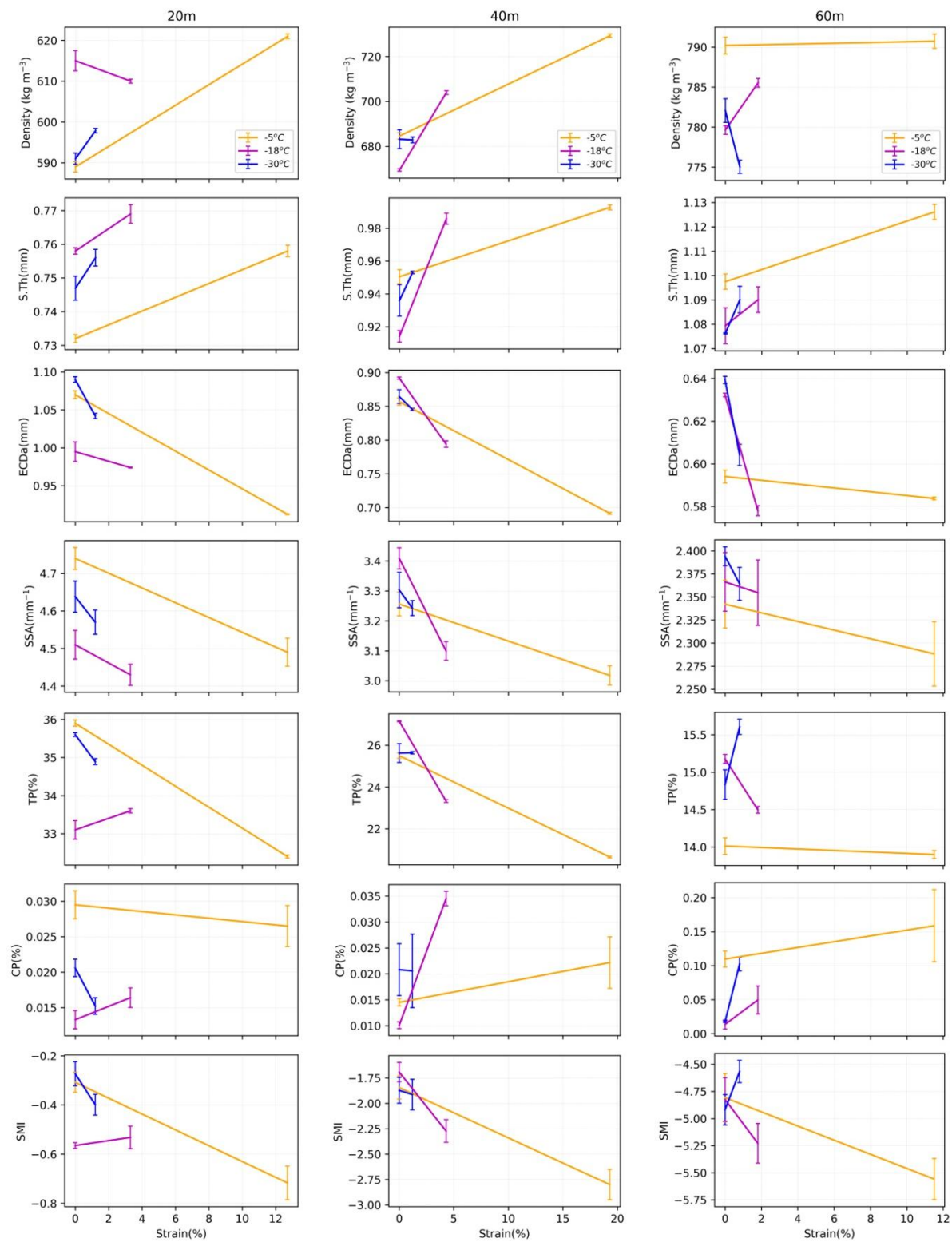
SSAs of $4.74 \pm 0.03 \text{ mm}^{-1}$, $4.51 \pm 0.04 \text{ mm}^{-1}$, and $4.64 \pm 0.04 \text{ mm}^{-1}$, respectively (**Figures 2–3; Table 1**). Here, despite the -18°C specimen having a higher density than the two others at -5°C and -30°C , is not possible to conclude that the sintering of firn is not directly related to the temperature. This is likely because a thermal equilibration period of two days in the absence of compression is too short to sufficiently exert the influence of temperature on firn sintering. The microstructural differences seen in these specimens more likely arose from the initial samples themselves, which were anisotropic and heterogeneous even if taken from the same depth, attributed to firn pre-deformation and partial annealing before experiments (Li and Baker, 2022a).

3.2 Microstructures after creep

The microstructural evolution is characterized by the microstructural parameters shown in **Figure 3**. The largest changes occurred in the -5°C specimens due to the higher temperature, i.e., the density, S.Th, and CP increased, while the ECDA, TP, SSA, and SMI decreased, indicative of consolidation of the firn after creep. It is important to note that for the 60 m sample tested at -5°C , there was no change in density, i.e., $790.2 \pm 1 \text{ kg m}^{-3}$ before creep vs. $790.7 \pm 0.9 \text{ kg m}^{-3}$ after creep, or TP, i.e., $14.0 \pm 0.1\%$ before creep vs. $13.9 \pm 0.1\%$ after creep. This lack of microstructural change is due to the high initial density, which was close to the firn pore close-off density of $\sim 830 \text{ kg m}^{-3}$. Thus, the creep of this sample may involve a transition from firn to

bubbly ice, as is also indicated by the increase in CP, which would have made it difficult to compress further. Intriguingly, some of the changes in microstructure observed in the micro-CT 3-D reconstructions from the specimens before and after creep, e.g. the distribution of ice-space, are indistinguishable in **Figure 2**. This is presumably due to the relatively large initial particle size, or from radial dilation exceeding the axial compression because of the small strains that occurred at relatively low temperatures.

One exception to the expected microstructural change after creep was the decrease of CP, which was likely due to the measurement uncertainty of the micro-CT (Burr et al., 2018), or radial expansion of the specimen during creep. Another exception was the decrease in density after creep for the -18°C specimen at 20 m and the -30°C specimen at 60 m, which arose due to a de-densification effect produced by temperature gradient metamorphism, as confirmed by the increase of both TP and S.Th (Li and Baker, 2022b). The thermal gradient appears to be associated with a fluctuation of 0.2°C around the test temperature, similar to temperature cycling occurred within firn (Mellor and Testa, 1969; Weertman, 1985), which stems from the thermometer's inherent accuracy as noted in Section 2.2 ($-5 \pm 0.2^{\circ}\text{C}$, $-18 \pm 0.2^{\circ}\text{C}$ and $-30 \pm 0.2^{\circ}\text{C}$). In the relatively simple deformation found at ice-sheet dome sites, such as Summit, there is no mechanism to decrease density during compression. At sites closer to the ice sheet margins, cracking due to extension of the ice may cause a localized decrease in density. The rate of firn densification should decrease with increasing depth at a given temperature, due to the decrease of effective stress with increasing depth (**Appendix A**). As a matter of fact, the density of the -5°C samples after creep increased by 32 kg m^{-3} , 44 kg m^{-3} , and 0.5 kg m^{-3} for the 20 m, 40

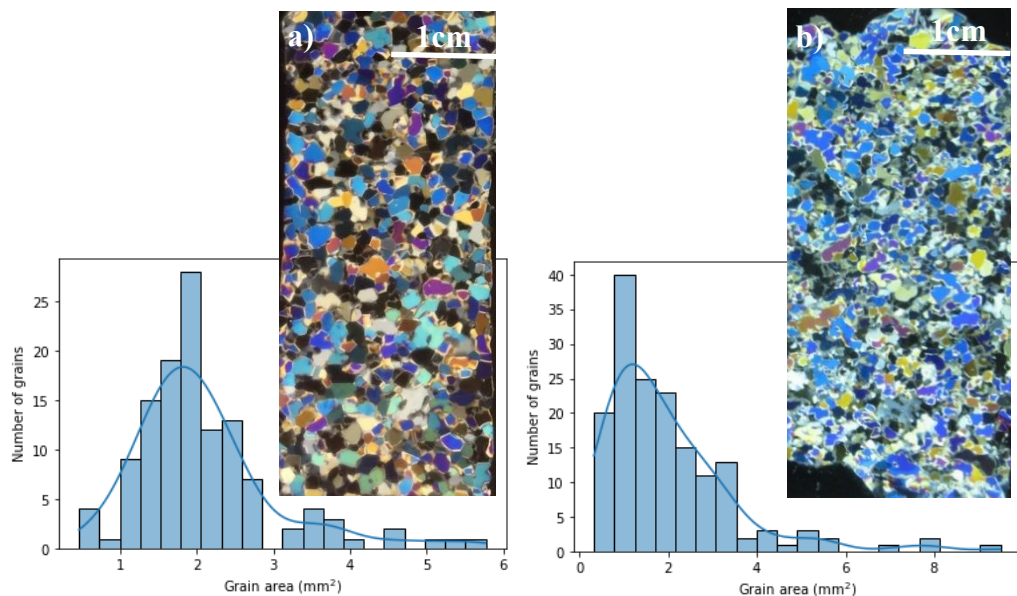


297 **Figure 3:** Density, structure thickness (S.Th), area-equivalent circle diameter (ECda), specific
298 surface area (SSA), total porosity (TP), closed porosity (CP), and structure model index (SMI) of

the firm samples before and after creep at three temperatures (orange, magenta, and blue lines) from depths of 20 m, 40 m and 60 m. Error bars indicate the variation of each microstructural parameter as derived from three different VOIs of the same sample.

m, and 60 m samples, respectively. The 44 kg m^{-3} unexpectedly outnumbers the 32 kg m^{-3} , implying that the densification of firm is also affected by other undetermined factors, e.g. the effect of inclusions, in addition to the stress and temperature.

Another way to investigate microstructure changes before and after creep tests is to compare their grain sizes using thin sections. As an example, **Figure 4** shows optical micrographs of thin sections made from the -5°C sample at 40 m before and after creep to a strain of 19.3%, where the significant reduction in grain size from $0.8 \pm 0.67 \text{ mm}$ to $0.5 \pm 0.32 \text{ mm}$ implies the occurrence of recrystallization during testing. However, it is also unclear at what strain recrystallization



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323 **Figure 4:** Optical micrographs of thin sections, and the distribution of grain sizes for the 40 m sample
324 at -5°C (a) before and (b) after creep (19.3% strain).

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327 was initiated in each test, as noted in Li and Baker (2022a). Recrystallization occurs frequently at
328 a temperature higher than the homologous temperatures of $0.9 T_m$. However, no evidence was
329 found for recrystallization after testing at the relatively cold -18°C and -30°C conditions,
330 probably due to the small creep strains at these relatively low temperatures. The creep
331 mechanisms for these samples, and whether the mechanisms were different at different
332 temperatures, could not be determined from the micro-CT-derived microstructural observations
333 alone, because the micro-CT can only capture the microstructure before and after creep. Instead,
334 plots of both strain vs. time and strain rate vs. strain can be used to elucidate the onset of
335 recrystallization during creep (Sections 3.3 and 3.4; Ogunmolasuyi, et al., 2023).

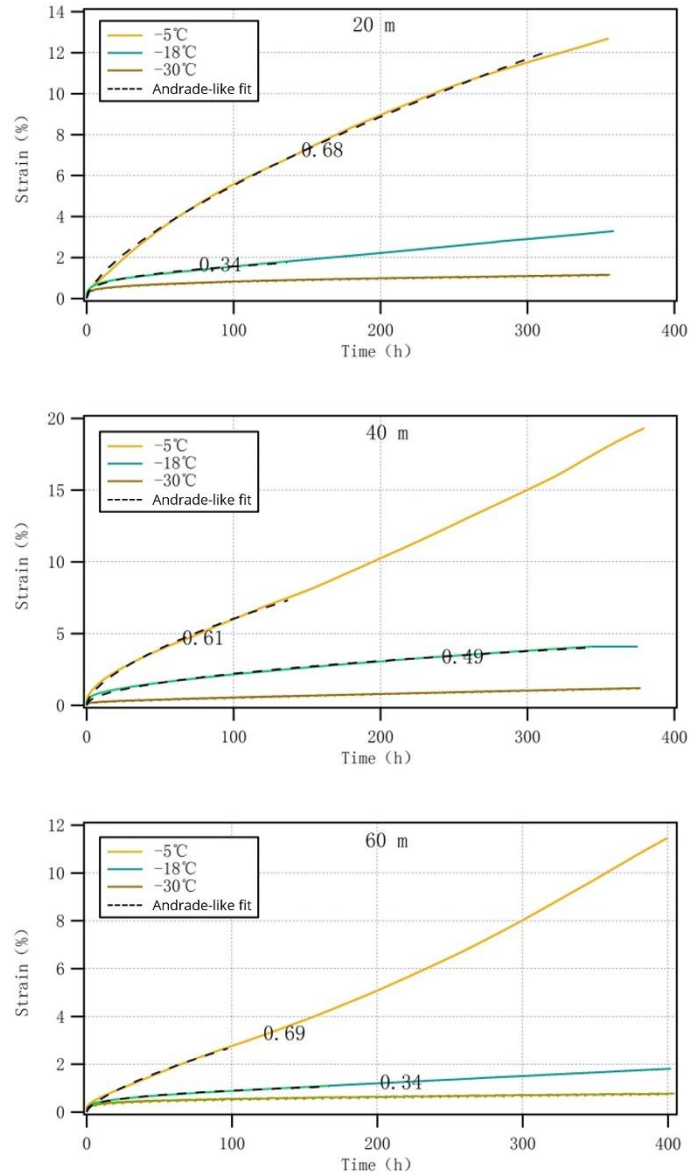
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337 *3.3 Relationship between strain and time*

338 **Figure 5** shows the strain vs. time creep curves. The specimens at -5°C at 20 m and -18°C at 20
339 m, 40 m, and 60 m, show decelerating transient creep and quasi-viscous steady-state creep, while
340 the specimens at -5°C at 40 m and 60 m show transient, secondary, and accelerating tertiary creep.
341 Note that the curves from the -30°C specimens are not easily interpreted due to a large amount of
342 noise arising from both the insufficient resolution of a linear voltage differential transducer (Li
343 and Baker, 2022a) and the very small strains. The transient creep stage may be caused by strain

hardening that occurs from the yield point to the ultimate strength (Glen, 1955; Jacka, 1984). The plastic deformation is accommodated by an increase in dislocation density through dislocation multiplication or the formation of new dislocations (Frost and Ashby, 1982; Duval et al., 1983; Ashby and Duval, 1985), which leads to an increase of the firm strength as the dislocations become pinned or tangled, and thus more difficult to move. The initial decrease of creep rate may also be related to the rearrangement of dislocations into a more stable pattern through a dragging mechanism (Weertman, 1983) for the -5°C specimens. The tertiary creep stage may be associated with strain softening deriving either from the thermally-activated processes at the high homologous temperature approaching the melting point of ice, or from recrystallization (Li and Baker, 2022a). Clearly, the creep rate of firm is sensitive to temperature under constant stress at a given depth, viz., the creep rate increases with increasing temperature (**Figure 5**). Incidentally, there is no evidence of the onset of recrystallization in the creep curves themselves despite the thin-section observation that -5°C specimens clearly underwent recrystallization during creep (Section 3.2).

A modified Andrade-like equation $\varepsilon = \beta t^k + \varepsilon_0$ in Li and Baker (2022a) was used to describe the transient creep behavior of the firm, in which the primary creep was well represented in black dashed lines on the creep curves in **Figure 5**. The time exponent k , derived from the above equation, ranges from 0.34–0.69: the data for the -30°C specimens are excluded since the noise in the results makes them uninterpretable. These k values are also smaller than those from monocrystalline and bicrystalline ice: 1.9 ± 0.5 , 1.5 ± 0.2 , and 1.3 ± 0.4 (Li and Baker, 2022a and references therein). We also note that the k values from the specimens at -5°C from 20–60 m



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Figure 5: Strain vs. time for firm specimens at -5°C (yellow lines), -18°C (blue lines), and -30°C (brown lines), from depths of 20 m (applied stress 0.21 MPa), 40 m (0.32 MPa) and 60 m (0.43 MPa). The black dashed curves represent fits to a modified Andrade-like equation with the time exponents indicated on the curves, if any. Note: The y-axis limits vary across the subfigures.

(0.68, 0.61, and 0.69), and at -18°C from 40 m (0.49) are greater than 0.33, while the k value from the -18°C specimens at 20 m (0.34) and 60 m (0.34) are close to 0.33 that is usually

obtained for full-density polycrystal ice. Interestingly, an evident relationship between the density of firm and the k values, regardless of the effect of stress (Li and Baker, 2022a) and temperature, remains unknown. A greater k value signifies faster deformation. The k values derived for firm are generally higher than those for polycrystalline ice, implying that the higher firm deformation rates compared to those of ice ($k = \sim 0.33$; Cuffey and Paterson, 2010, and references therein) are likely related to the fewer grain-boundary constraints with more void space in firm (Li and Baker, 2022a; Li, 2023b). Clearly, the above k values, which increased with increasing temperature (**Figure 5**), indicate that deformation is easier because of the lower viscosity at the higher temperature. Thus, k seems to be a state variable with respect to temperature. In addition, k values greater than 0.33 may be related to the decrease of viscosity of the firm specimens (Freitag et al., 2002; Fujita et al., 2014). k values lower than 0.33 observed under constant load and temperature occurred at relatively low effective stresses (Li and Baker, 2022a). The identified trend of steadily declining k values across the temperature range of -5°C to -18°C , however, represents a significant gap in our current understanding, necessitating a dedicated investigation into the microstructural or metamorphic causes. Alternatively, the enhanced cohesion strength in the firm, which resulted from both the ice matrix with higher purity and the stronger bond connection of inter-grains, increases the viscosity of test samples and lowers the k value to less than 0.33.

3.4 Relationship of strain rate to strain

Figure 6 shows log strain rate vs. strain plots from all the -5°C and -18°C specimens; the -30°C samples are excluded due to noise. The evolution of the strain rate is characterized more clearly in **Figure 6** than in **Figure 5**. Clearly, the strain rate is also a state variable of temperature, where the

strain rate increases with increasing temperature for a given strain at a given depth (**Figure 6**; **Table 2**). The strain rate minimum at the secondary creep stage (SRmin) and the strain at the

Table 2. Observed and inferred strain rate minima and strains observed at the strain rate minima.

20 m	SRmin s ⁻¹	PC1-SRmin s ⁻¹	PC2-SRmin s ⁻¹	PC3-SRmin s ⁻¹	Strain %
–5°C	5.53×10^{-6}	<i>5.53×10^{-6}</i>	<i>1.68×10^{-6}</i>	<i>2.56×10^{-7}</i>	11.8
–18°C	1.36×10^{-6}	<i>1.36×10^{-6}</i>	<i>2.29×10^{-7}</i>	<i>2.45×10^{-8}</i>	1.81–2.9
–30°C(U)	–	7.14×10^{-7}	2.17×10^{-7}	3.3×10^{-8}	–
–30°C(L)	–	3.16×10^{-8}	9.6×10^{-9}	1.46×10^{-9}	–

40 m	SRmin s ⁻¹	PC1-SRmin s ⁻¹	PC2-SRmin s ⁻¹	PC3-SRmin s ⁻¹	Strain %
–5°C	1.03×10^{-5}	3.39×10^{-5}	<i>1.03×10^{-5}</i>	<i>1.57×10^{-6}</i>	7.5
–18°C	1.4×10^{-6}	8.32×10^{-6}	<i>1.40×10^{-6}</i>	<i>1.5×10^{-7}</i>	4.1
–30°C(U)	–	4.37×10^{-6}	1.33×10^{-6}	2.03×10^{-7}	–
–30°C(L)	–	1.94×10^{-7}	5.88×10^{-8}	8.97×10^{-9}	–

60 m	SRmin s ⁻¹	PC1-SRmin s ⁻¹	PC2-SRmin s ⁻¹	PC3-SRmin s ⁻¹	Strain %
–5°C	5.59×10^{-6}	1.21×10^{-4}	3.67×10^{-5}	<i>5.59×10^{-6}</i>	2.7
–18°C	5.33×10^{-7}	2.96×10^{-5}	4.99×10^{-6}	<i>5.33×10^{-7}</i>	1.1–1.8
–30°C(U)	–	1.56×10^{-5}	4.74×10^{-6}	7.21×10^{-7}	–
–30°C(L)	–	6.91×10^{-7}	2.1×10^{-7}	3.19×10^{-8}	–

The SRmin without the prefix is the observed values during creep, while the SRmin with a prefix is the inferred values. Note that PC-SRmin is the abbreviation of the post-calibration SRmin, and that –30°C(U) and –30°C(L) indicate the upper and lower bound from the –30°C samples from 44.8 kJ mol⁻¹ and 113 kJ mol⁻¹, respectively. PC1-SRmin, PC2-SRmin, and PC3-SRmin are described in **Appendix B**. The symbol – indicates the unavailable values of SRmin and the strain value at the SRmin observed during creep. For the italics highlighted, see **Appendix B**.

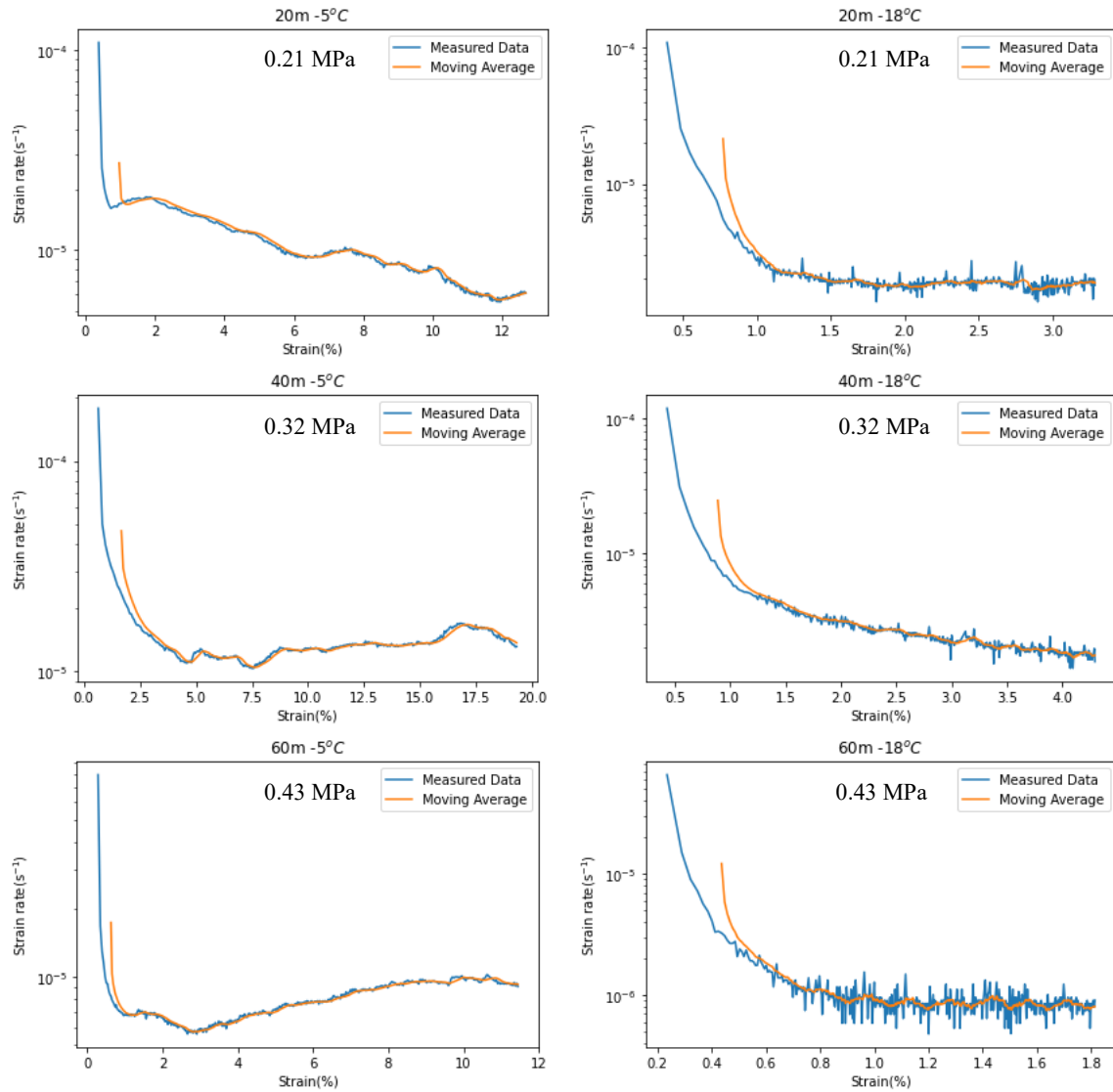


Figure 6: Log strain rate vs. strain from the firn specimens at temperatures of -5°C and -18°C from depths of 20 m (applied stress 0.21 MPa), 40 m (0.32MPa) and 60 m (0.43MPa). Samples from -30°C are not shown due to the very large noise. The blue lines represent discrete strain rates, which are calculated by extracting the strain data hourly, while the orange lines represent a moving average of 15 moving windows with respect to the strain.

SRmin for all the -5°C and -18°C specimens are shown in **Figure 6** and **Table 2**. The SRmin was reached at a strain of 11.8%, 7.5% and 2.7% for the -5°C specimens from depths of 20 m, 40 m, and 60 m, respectively, consistent with strains at the SRmin decreasing with increasing depth at a given temperature in **Figure 8** and **Table 4** in Li and Baker (2022a). For the -18°C specimens, the SRmin occurred over a range of strains from 1.81–2.9% at 20 m, at a fixed strain of 4.1% at 40 m, and at a strain oscillating between 1.1 and 1.8% at 60 m. These values of strain at different SRmin values are different from those usually observed at strains of 0.5–3% for fully-dense ice (Cuffey and Paterson, 2010, and references therein), implying different mechanical behavior between firn and pure ice (Duval, 1981; Mellor and Cole, 1983; Jacka, 1984; Li et al., 1996; Jacka and Li, 2000; Song et al., 2005, 2008; Cuffey and Paterson, 2010). Overall, the strain at the SRmin is greater with lower density and higher temperature, e.g. 11.8% strain from the -5°C specimens at 20 m, and 4.1% strain from the -18°C specimens at 40 m. This is likely due to the effect of strain hardening on density and temperature (Li, 2023b). Additionally, tertiary creep is observed during both quasi-steady state deformation, particularly in the -5°C specimens at depths of 40 m and 60 m, and in the ascending stage, as seen in the -5°C and -18°C specimens at 20 m, along with the -18°C specimen at 40 m. This mechanical behavior is facilitated by lower firn density, increased effective stress, and elevated creep temperatures. For instance, in the -5°C specimens at 20 m, strain softening primarily results from recrystallization (Duval, 1981; Jacka, 1984; Jacka and Li, 2000; Song et al., 2005; Faria et al., 2014). Also, the activation of easy slip systems contributes to this process (Jonas and Muller, 1969; Duval and Montagnat, 2002; Alley et al., 2005; Horhold et al., 2012; Fujita et al., 2014; Eichler et al., 2017; Vedrine et al., 2025). It is noteworthy that Jacka and Li (1994) observed that steady-state tertiary ice creep, which is marked

444 by stable grain size, is influenced more by applied stresses than by temperature. This finding
445 suggests that there exists a balance between the activation energies required for grain growth and
446 subdivision at a specific temperature.

447

448 3.5 Apparent activation energy for creep

449 Experimental observations of the SRmin are limited, as they only occurred for the -5°C and at $-$
450 18°C specimens at each depth (**Table 2**). It is hard to achieve the SRmin for all firm specimens in
451 laboratory environments (Landauer, 1958), especially under low temperatures and stresses such as
452 those from the -30°C specimens in this work. To this end, we offer the various possibilities of the
453 SRmin using the evidence we have. The value of the apparent activation energy of creep, Q_c (kJ
454 mol^{-1}), is equal to the slope of a line fitted $\ln \dot{\epsilon}$ versus $1/T$ as did in Goldsby & Kohlstedt (1997;
455 2001), using the Arrhenius relation $\dot{\epsilon} = B\sigma^n \exp(-\frac{Q_c}{RT})$, where $\dot{\epsilon}$ (s^{-1}) is the strain rate, B (s^{-1}
456 Pa^{-n}) is the material parameter, σ (MPa) is the applied stress, n is the creep (stress) exponent, R
457 ($8.314 \text{ J mol}^{-1} \text{ K}^{-1}$) is the gas constant, and T (K) is Kelvin temperature. It is noted that under
458 constant-stress conditions, the value of the stress exponent n influences the pre-exponential factor
459 B but does not affect the slope of the Arrhenius plot and therefore the derived activation energy Q_c .
460 First, the estimation of Q_c is based on *only* two SRmin values from the -5°C and -18°C samples
461 at each depth (**Table 2**). Glen-King's model $\dot{\epsilon} = A \exp(-Q_c/RT) = B\sigma^n \exp(-Q_c/RT)$ treats the
462 pre-factor A , material parameter B , and stress exponent n as constants (Glen, 1955; Goldsby and
463 Kohlstedt, 2001). This simplification is valid by using the unifying concept of normalized
464 effective stress. The effective stress captures the complex multi-physical behavior of the
465 two-phase ice-air system, accounting for: 1) The incompressibility of individual ice grains versus

466 the compressibility of the porous ice skeleton, 2) The coupled flow of ice and air; and 3) The
 467 interplay between different strain components (axial, radial, volumetric, and true). This
 468 framework is grounded in the principles of poromechanics, originally developed for soils and
 469 later applied to snow and ice (Gubler, 1978; Hansen and Brown, 1988; Mahajan and Brown, 1993;
 470 Chen and Chen, 1997; Lade and deBoer, 1997; Ehlers, 2002; Khalili et al., 2004; Gray and
 471 Schrefler, 2007; daSilva et al., 2008; Nuth and Laloui, 2008). The variability in density for the
 472 samples from 20-m depth on the mechanical behavior are negligible due to a small difference (up
 473 to ~4%), between samples, which falls within an acceptable error range in previous studies. This
 474 is likely related to multiple factors, including the intrinsic properties of the samples, e.g.
 475 inclusions (impurities, dust, bubbles, clathrate hydrates), the effects of deformation and partial
 476 annealing of firn due to stress distribution and temperature changes during drilling, extraction,
 477 transportation, or storage, and the fact that the samples are taken from adjacent parts of the core,
 478 and might capture heterogeneous density layers, as well as potential measurement errors
 479 associated with the equipment used. The Q_c values from the 20 m, 40 m, and 60 m specimens
 480 were calculated to be 61.4 kJ mol⁻¹, 87.3 kJ mol⁻¹, and 102.8 kJ mol⁻¹, respectively (**Figure 7**).
 481 Based on the three SRmin from the -5°C and -18°C samples at 60 m in this work, and from -
 482 10°C samples at 60 m in Li and Baker (2022a), a Q_c value for the 60 m specimen was calculated
 483 to be 100.7 kJ mol⁻¹. To see whether or not these above Q_c values are reliable, we estimated the
 484 activation energy of grain-boundary diffusion/viscosity, Q_{gbd} (kJ mol⁻¹), using the relation
 485 $K = (D_t^2 - D_0^2)/t = k \exp(-Q_{\text{gbd}}/RT)$, in an alternative form of $Q_{\text{gbd}} = -R[\partial \ln K / \partial (1/T)]$,
 486 where K is the observed rate of grain growth (mm² a⁻¹), D_0^2 and D_t^2 are the measured mean
 487 grain area (mm²) in a firn sample at the onset of the creep ($t = 0$), and at the end time of the creep

(t-year), and k is a constant grain growth factor. The grain growth rates are plotted on a logarithmic scale against the reciprocal of T (**Figure 7**). For changes in grain size from the related specimens before and after creep see **Table 3**. Correspondingly, the Q_{gbd} values calculated were 41.4 kJ mol⁻¹, 40.8 kJ mol⁻¹, and 40.9 kJ mol⁻¹ for the specimens at 20 m, 40 m, and 60 m, respectively. These Q_{gbd} values are comparable to the values of 40.6 kJ mol⁻¹ obtained in laboratory experiments on polycrystalline ice (Jumawan, 1972), and 42.4 kJ mol⁻¹ from 13 polar firn cores (Cuffey and Paterson, 2010) for grain-boundary self-diffusion of polycrystalline ice. Further, the ratio of Q_{gbd}/Q_c is 0.67, 0.47, and 0.4 for the 20 m, 40 m, and 60 m specimens, respectively. We noted that the ratio of 0.67 for Q_{gbd}/Q_c was recommended by Hobbs (1974) and Cuffey and Paterson (2010). The Q_c values calculated using the Arrhenius relation for the 40 m and 60 m specimens are likely greater than the actual values, and hence are seemingly less reliable. There is little difference between the two-SRmin-derived Q_c value (102.8 kJ mol⁻¹) and the three-SRmin-derived Q_c value (100.7 kJ mol⁻¹), implying that these two avenues for calculating Q_c have equal utility. Moreover, the above Q_{gbd} values are lower than the 48.6 kJ mol⁻¹ that was inferred by the grain growth rate for firn samples with densities ranging from 320–650 kg m⁻³ from cores drilled at the South Pole, Antarctic (Gow, 1969), which makes a ratio of 0.67 for Q_{gbd}/Q_c an unreliable sole-criterion. In short, it is difficult to assess the reliability of both Q_c and Q_{gbd} , as discussed above due to their scatter and debates in the current literature. Thus, these Q_c values estimated in this work, ranging from 61.4–102.8 kJ mol⁻¹, are reasonable, aligning with the literature range of 44.8–113 kJ mol⁻¹ (**Table 4**).

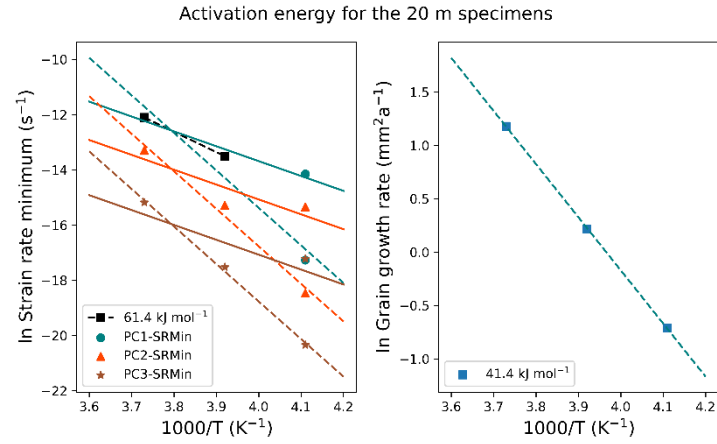
Table 3. Grain area (mm^2) measured from optical thin sections for samples at -5°C , -18°C , and -30°C from depths of 20 m, 40 m, and 60 m before and after creep.

Depth	20 m		40 m		60 m	
T/ $^\circ\text{C}$	Before	After	Before	After	Before	After
-5	0.29 ± 0.25	0.42 ± 0.28	0.53 ± 0.32	0.79 ± 0.67	0.78 ± 0.67	0.97 ± 0.8
-18	0.29 ± 0.25	0.34 ± 0.2	0.53 ± 0.32	0.7 ± 0.42	0.78 ± 0.67	0.9 ± 0.59
-30	0.29 ± 0.25	0.31 ± 0.17	0.53 ± 0.32	0.57 ± 0.34	0.78 ± 0.67	0.81 ± 0.56

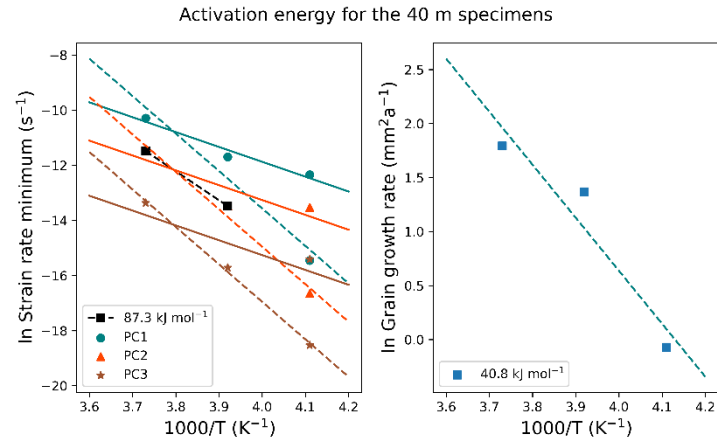
A great challenge is the estimation of the Q_c using the SRmin including the -30°C specimens, whose SRmin shows high variability due to the extraordinarily slow strain rate at low temperatures. This difficulty cannot be resolved by extrapolating experimental data (Sinha, 1978; Hooke et al., 1980), e.g. the use of Andrade's law (Glen, 1955). Instead, we turned our focus to studying the relationship between the SRmin and temperature by constraining our data in a wide range of Q_c values reported in existing literature presented in **Table 4**. Clearly, there is a larger scatter of Q_c values for firn than for ice. The increase of Q_c from mono-crystalline and bi-crystalline to polycrystalline ice implies that the greater the reduction in the constraint from grain boundaries, the greater is Q_c . Alternatively, firn creep is easier than that of polycrystalline ice due to either the easier sliding of grains in firn along more directions in the more porous and heterogeneous structure (Section 3.3), or the decrease of viscosity associated with inclusions (e.g.

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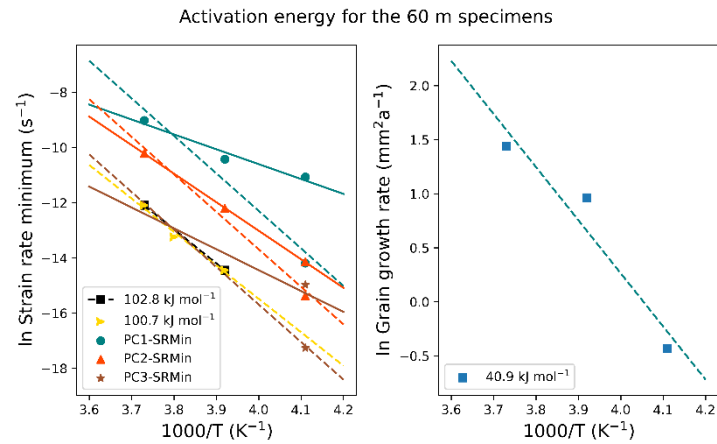
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530 **Figure 7:** Arrhenius plots to estimate the apparent activation energy of creep (Q_c ; left panel) and
531 the apparent activation energy of grain-boundary diffusion (Q_{gbd} ; right panel) from the firm
532 specimens noted. The teal, orange, and brown solid lines are the upper bound (44.8 kJ mol⁻¹) of
533 PC1-SRmin, PC2-SRmin, and PC3-SRmin, respectively, while the teal, orange, and brown dashed

lines are the lower bound (113 kJ mol⁻¹) of PC1-SRmin, PC2-SRmin, and PC3-SRmin, respectively (Table 2). The teal circles, the orange triangles, and the brown stars are the data in Table 2. The black dashed lines are from *only* two SRmins at -5°C and -18°C (the black squares are the data measured), whose Q_c is indicated in each subfigure. The yellow dashed line is from the three SRmins at -5°C, -18°C in this work, and -10 °C from Li and Baker (2022a) (the yellow triangles are the measured data), whose Q_c is 110.7 kJ mol⁻¹. The blue dashed lines (right panel) are from grain growth rate at three temperatures (the blue squares are the observed data), whose Q_{gbd} is indicated in each subfigure.

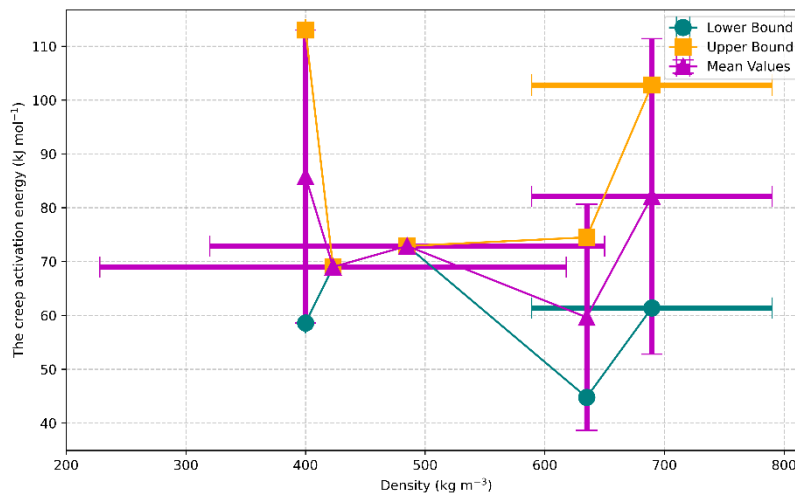
Table 4. Apparent activation energy for the creep of firn and ice, Q_c , reported in literature.

Q_c kJ mol ⁻¹	Sample	Density kg m ⁻³	Temperature °C	Methods	Source
58.6–113	Firn	~400	[-13.6, -3.6]	Uniaxial/Hydrostatic Compression	Landauer (1958)
44.8–74.5	Firn/Bubbly Ice	440–830	[-34.5, -0.5]	Uniaxial Unconfined Compression	Mellor and Smith (1966)
54	Firn/Bubbly Ice	Undetail ed	[-28, -16]	Shear Deformation of Boreholes	Paterson (1977)
~72.9	Firn	320–650	Unnecessary	Grain Growth Rate	Gow (1969)
69 ± 5	Firn	423 ± 8	[-19, -11]	Triaxial Compression	Scapozza and Bartelt (2003)
~60	Artificial/Natural Ice (South Pole)	~917	-15	Torsion Creep Test	Pimienta and Duval (1987)
61	Polycrystalline Ice	~917	-9.6	Hydrostatic Pressure	Duval et al. (1983)
78	Monocrystalline Ice	~917	[-30, -4]	Derived from Bicrystal Ice	Homer and Glen (1978)
75	Ice Bicrystal	~917	[-30, -4]	Tensile Test Parallel to Grain-boundary	Homer and Glen (1978)

546

547 Baker and Gerberich, 1979; Goodman et al., 1981) that facilitate the intra- and inter-grain sliding
548 (Salamatin et al., 2009). In principle, Q_c of firm should exceed that for polycrystalline ice.
549 Intriguingly, some reported Q_c values from firm are less than that for ice, meaning the degree of
550 spatial freedom in the ice-matrix is limited by the topological structure of the firm (Liu et al.,
551 2022). Incidentally, the effective stress of porous materials is determined by not only its porosity,
552 but also other factors, e.g. the microstructural topology (Liu et al., 2022) and the impurity types
553 and concentrations in the firm. However, this issue is beyond the scope of this work. In seeking a
554 conclusion, we evaluated the dependence of creep activation energy on firm density. The data
555 indicate no discernible relationship between these two parameters (**Fig. 8**). In summary, a Q_c for
556 firm, which ranges from 44.8–113 kJ mol⁻¹, is plausible due to the intrinsic nature of natural firm
557 that has a far more complicated and changeable microstructure than ice.

558



559

560 **Figure 8:** Plots of the creep activation energy vs. firm densities. For each density, three values are
561 shown: the lower bound (minimum activation energy, teal), the upper bound (maximum activation
562 energy, orange), and the mean value (magenta). Error bars represent the standard deviation of the

mean. Data are sourced from Table 4 and the present study.

The value of the stress exponent n is determined by plotting the line fit the logarithm relation of the steady-state strain rate, $\dot{\epsilon}$, versus the effective stress, σ , and is, thus, the slope of this line from the measured SRmins (**Table 2**). We determined stress exponent (n) values of approximately 0.1 and -1.2 for the -5°C and -18°C samples based on observed data, respectively. This result directly contradicts the value of $n \approx 4.3$ reported from the same Greenland firn core by Li and Baker (2022a). Further, these values fall entirely outside the established range of ~ 1 to ~ 7.5 (mean $\sim 4.25 \pm 3.25$) documented across decades of ice mechanics literature (Glen, 1955; Hansen and Landauer, 1958; Butkovich and Landauer 1960; Kamb, 1961; Paterson and Savage, 1963; Higashi et al, 1965; Mellor and Testa, 1969; Raymond, 1973; Hooke, 1981; Thomas et al., 1980; Duval et al., 1983; Weertman, 1983,1985; Azuma and Higashi, 1984; Pimienta and Duval, 1987; Budd and Jacka, 1989; Jacka and Li, 1994; Goldsby and Kohlstedt, 2001; Bindshadler et al., 2003; Cuffey, 2006; Chandler et al. 2008; Cuffey and Kavanaugh, 2011; McCarthy et al., 2017; Millstein et al., 2022; Colgan et al., 2023; Li, 2025). The wide range of reported n -values is governed by a complex interplay of deformation mechanisms—including grain boundary sliding, diffusion (lattice and grain boundary), and dislocation processes, e.g. hard-slip-dominated, dislocation-accommodated grain boundary sliding, and grain boundary sliding-limited basal dislocation—across varying stresses, temperatures, crystallographic fabrics, impurity contents, and grain-size-to-sample-size ratios. We attribute the significant discrepancy in these findings to the experimental conditions. The lower temperatures used (down to -30°C) induce slower strain rates,

585 which prevented the tests from reaching a critical strain rate minimum (SRMin). Therefore, to
 586 accurately estimate the activation energy for deformation, it is necessary first to calibrate the
 587 SRMin value for all noised samples. A constant stress exponent value of $n \approx 4.3$ (Li and Baker,
 588 2022a) was used to compute the activation energy. This necessary simplification—an
 589 acknowledgement of current methodological limitations rather than a dismissal of the underlying
 590 physics—introduces a key uncertainty that highlights the need for future advancements in
 591 observational methodology within firm research. To proceed, the post-calibration SRmins for the –
 592 5°C and –18°C samples are highlighted in **Table 2** (see **Appendix B** in detail). It is important to
 593 note that the stress exponent does not depend on the density of the tested samples, thereby
 594 negating any basis for discussing a relationship between the stress exponent and sample density.
 595 Instead, variations in stress corresponding to density variations are manifested in the strain rate,
 596 ensuring that the derivation of the stress exponent and activation energy remains consistent. From
 597 here on we only discuss the applied stress since there is little difference between the effective
 598 stress and applied stress for calculating the stress exponent (Li and Baker, 2022a). Based on both
 599 the reported range of Q_c and the two observed SRmins at –5°C and –18°C, the SRmins for the
 600 –30°C samples are inferred (**Table 2**), using the Arrhenius relation. Also, based on both the
 601 observed and inferred SRmins with the upper and lower bounds (**Table 2**), a series of fitted
 602 functions are then found between the SRmin and the reciprocal of the temperature (°C), $1/T_c$:

603

604 20-m samples:

$$\left\{ \begin{array}{l} \text{SRMin} = -3 \times 10^{-5} / T_c - 7 \times 10^{-7} [R^2 = 0.988; \text{PC 1(L 20)}] \\ \text{SRMin} = -3 \times 10^{-5} / T_c - 2 \times 10^{-7} [R^2 = 1; \text{PC 1(U 20)}] \\ \text{SRMin} = -1 \times 10^{-5} / T_c - 3 \times 10^{-7} [R^2 = 1; \text{PC 2(L 20)}] \\ \text{SRMin} = -9 \times 10^{-6} / T_c - 2 \times 10^{-7} [R^2 = 0.987; \text{PC 2(U 20)}] \\ \text{SRMin} = -2 \times 10^{-6} / T_c - 6 \times 10^{-8} [R^2 = 0.998; \text{PC 3(L 20)}] \\ \text{SRMin} = -1 \times 10^{-6} / T_c - 3 \times 10^{-8} [R^2 = 0.976; \text{PC 3(U 20)}] \end{array} \right\},$$

40-m samples:

$$\left\{ \begin{array}{l} \text{SRMin} = -2 \times 10^{-4} / T_c - 4 \times 10^{-6} [R^2 = 0.988; \text{PC 1(L 40)}] \\ \text{SRMin} = -2 \times 10^{-4} / T_c - 2 \times 10^{-6} [R^2 = 1; \text{PC 1(U 40)}] \\ \text{SRMin} = -6 \times 10^{-5} / T_c - 2 \times 10^{-6} [R^2 = 1; \text{PC 2(L 40)}] \\ \text{SRMin} = -6 \times 10^{-5} / T_c - 1 \times 10^{-6} [R^2 = 0.987; \text{PC 2(U 40)}] \\ \text{SRMin} = -1 \times 10^{-5} / T_c - 3 \times 10^{-7} [R^2 = 0.998; \text{PC 3(L 40)}] \\ \text{SRMin} = -9 \times 10^{-6} / T_c - 2 \times 10^{-7} [R^2 = 0.976; \text{PC 3(U 40)}] \end{array} \right\},$$

60-m samples:

$$\left\{ \begin{array}{l} \text{SRMin} = -7 \times 10^{-4} / T_c - 2 \times 10^{-5} [R^2 = 0.988; \text{PC 1(L 60)}] \\ \text{SRMin} = -6 \times 10^{-4} / T_c - 6 \times 10^{-6} [R^2 = 1; \text{PC 1(U 60)}] \\ \text{SRMin} = -2 \times 10^{-4} / T_c - 7 \times 10^{-6} [R^2 = 1; \text{PC 2(L 60)}] \\ \text{SRMin} = -2 \times 10^{-4} / T_c - 4 \times 10^{-6} [R^2 = 0.987; \text{PC 2(U 60)}] \\ \text{SRMin} = -3 \times 10^{-5} / T_c - 1 \times 10^{-6} [R^2 = 0.998; \text{PC 3(L 60)}] \\ \text{SRMin} = -3 \times 10^{-5} / T_c - 7 \times 10^{-7} [R^2 = 0.976; \text{PC 3(U 60)}] \end{array} \right\},$$

where PC1(L20) and PC1(U20) indicate the lower and upper bound values of the post-calibration SRmins from the 20 m samples (**Table 1**), and other symbols are similarly formatted, e.g. PC1(L40), PC1(U40), PC1(L60), PC1(U60), and so on. These relationships are plotted in **Figure 8**, where the SRmin vs. $1/T_c$ plots from the three depths are almost the same shape, implying that the SRmin is dependent on the temperature at a constant stress. It is important to note that the average (minimum) strain rate for the secondary creep stage for a given temperature increases with increasing depth/density of the samples (**Figure 8; Table 2**). This is opposite to a decrease of

618 the SRmin at a fixed stress and temperature in **Figure 8** and **Table 4** in Li and Baker (2022a).
619 These changes in SRmin are irrespective of the stress (**Appendix A**). The temperature plays a
620 predominant role during firn creep for a given density of sample at a constant stress. An
621 interesting question on firn creep at a specific temperature is whether the SRmin slows down or
622 speeds up with decreasing density of firn. Certainly, natural firn samples raise the complexity in
623 interpreting the firn creep due to the influences both from inclusions (Li and Baker, 2022a and
624 references therein; Li, 2024), and from the topology of the microstructures (Liu et al., 2022). In
625 addition, there is a broad spread of the SRmin at each depth, in which the SRmin varies by several
626 times, even one order of magnitude or more between the different possibilities of post-calibration
627 SRmins (**Figure 8**), implying that the microstructure of the sample significantly influences the

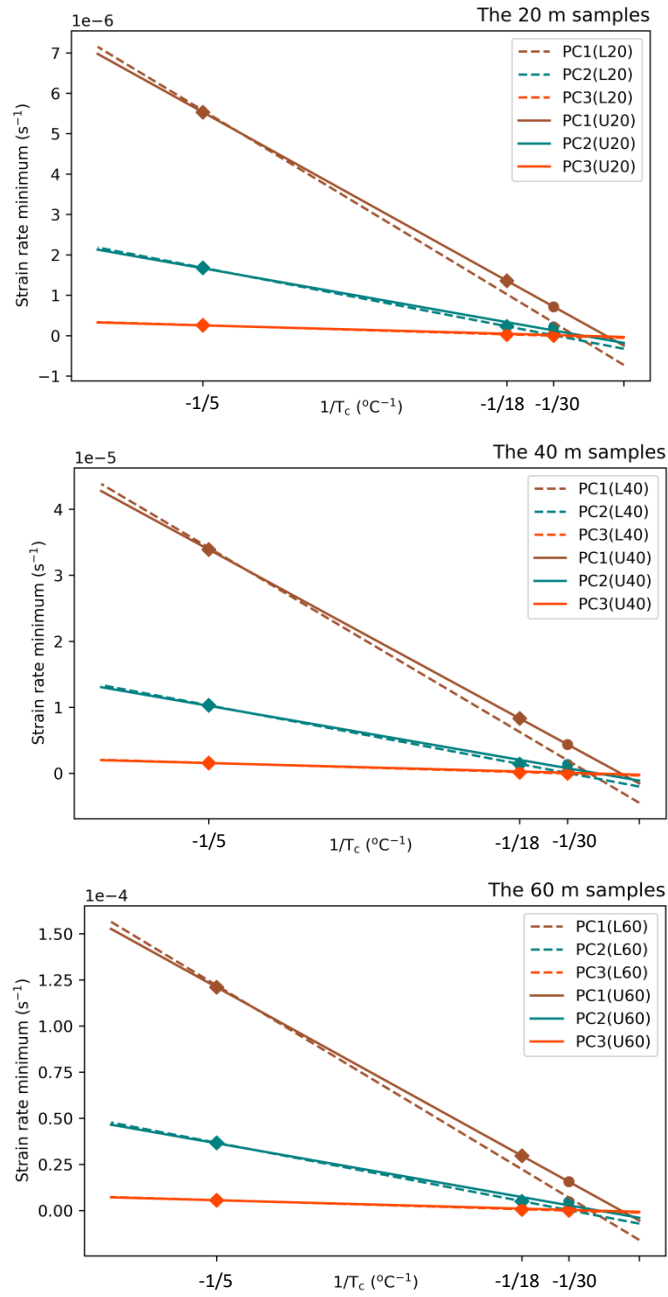


Figure 9: Plots of the strain rate minimum versus the reciprocal of temperature. PC1(L20) and PC1(U20) indicates the lower and upper bound, respectively, from the 20 m samples via PC1 as noted in Table 2, and so on. The circles indicate the upper bound data measured and inferred, while the squares indicate the lower bound data. The dashed line is the fit from the lower bound, while the solid line is the fit from the upper bound.

650

651 process of the creep of firn. Moreover, it is hard to generalize a universal formula for predicting
652 the SRmin at temperatures below -30°C , where the SRmins becomes negative (**Figure 8**). Thus,
653 there is a need for an in-depth understanding of the polar firn creep behavior in secondary creep
654 stage.

655

656 To illustrate the differences between the Q_c values calculated from PC1-SRmin, PC2-SRmin,
657 and PC3-SRmin, we have plotted them in **Figure 7**. Interestingly, the Arrhenius plots of the
658 natural logarithm of strain rate with $1000/T$ (**Figure 7**) are similar to those observed by Glen
659 (1955) and Homer and Glen (1978), implying that there is no significant difference in the creep
660 mechanism for a temperature range of -30°C to -5°C (Glen, 1955; Homer and Glen, 1978),
661 where both diffusion via grain-boundary, vacancy or interstitial defects (Barnes et al., 1971;
662 Brown and George, 1996; Nasello et al., 2005; Li and Baker, 2022b), and dislocations contribute
663 to the creep of polar firn.

664

665 **4. Conclusions**

666 Constant-load creep tests were performed on three cylindrical specimens tested from depths of 20
667 m (applied stress 0.21 MPa), 40 m (0.32 MPa) and 60 m (0.43 MPa) at temperatures of $-5 \pm$
668 0.2°C , $-18 \pm 0.2^{\circ}\text{C}$, and $-30 \pm 0.2^{\circ}\text{C}$ from a firn core extracted at Summit, Greenland in June
669 2017. The microstructures were characterized before and after creep testing using the micro-CT
670 and thin sections viewed between optical crossed polarizers. It was found that:

671

1. Microstructural parameters measured using the micro-CT show that the polar firn densified during the creep compression (e.g. from 685 to 729 kg m⁻³ for the 40 m specimen at -5°C), viz., the TP (from 25.5 to 20.7%), the ECDA (from 0.86 to 0.69 mm), the SSA (from 3.26 to 3.02 mm⁻¹), and the SMI (from -1.85 to -2.8) decreased, while the S.Th (from 0.95 to 0.99 mm) and the CP (from 0.01 to 0.02%) increased. Anomalies in the microstructures, especially at low temperatures of -18°C and -30°C, are likely due to metamorphism under temperature gradients, the radial dilation effect during firn deformation, the measurement uncertainty of the micro-CT, or the anisotropy and the heterogeneity of natural firn.
2. The transient creep behavior of firn at constant stress and different temperatures obeys an Andrade-like law, but, the time exponent k of 0.34–0.69 is greater than the 0.33 found for ice. This is due to fewer grain-boundary constraints in porous firn than in ice.
3. The secondary creep behavior of firn at constant stress and different temperatures presented here shows that the strain at the SRmin increases with decreasing firn density and increasing creep temperature. In particular, low-density firn during creep at high temperatures shows that the strain at the SRmin, e.g. 11.8% and 7.5% respectively from the 20 m and 40 m specimens at -5°C, is greater than the strain of 3%, which is the maximum found at the SRMin of ice.
4. The tertiary creep behavior of firn at constant stress and different temperatures is more easily observed from lower-density specimens at greater effective stresses and higher creep temperatures. The strain softening in tertiary creep is primarily due to recrystallization.
5. The apparent activation energy for the firn creep has a wide range of 61.4–102.8 kJ mol⁻¹ because the grains in firn slide more easily along more directions in the more porous and heterogeneous structure, the enhanced fluidity from inclusions, and the topological structure

of the firn. In addition, the SR_{min} is a function of the temperature, depending on the microstructure of firn and the inclusion content. The predicted SR_{min} increases with increasing firn density at a given temperature and is independent of the effective stress. Lastly, there is no significant difference in the creep mechanism at temperatures ranging from $-30^{\circ}C$ to $-5^{\circ}C$.

The creep of polar firn behaves differently from full-density ice, implying that firn densification is an indispensable process in fully understanding the transformation of snowfall to ice in the polar regions. Observed firn deformation indicates that temperature plays a determined role in firn densification. Thereby, it will be helpful to bridge a gap between the firn temperature and the climate of the past for reconstructing paleoclimate. Also, it will be helpful to apply a confining load to investigate the microstructure of the creep of polar firn with smaller initial particle sizes at low temperatures using the micro-CT. Further studies of interest are to investigate the quantitative relationship between the microstructural parameters and the mechanical behavior of polar firn, and when the onset of recrystallization occurs during creep, as well as verify the SR_{min} predicted by the relationship of SR_{min} vs. temperature from the firn specimens at more extensive ranges of stresses and temperatures.

711 **Appendix A:** Hydrostatic pressure, the applied stress, and the effective stress

712 The hydrostatic pressure, p , which varies with temperature, along with the cohesion of the ice and
713 the friction angle between snow particles, plays a significant role in determining the apparent
714 activation energy and, consequently, the strength of the ice (Fish, 1991). It was calculated from
715 the overburden pressure of snow, i.e. $p = \bar{\rho}_f g h$, where $\bar{\rho}_f$ is the average firm density above the
716 depth of interest, h , and g is the acceleration of gravity. At Summit, p at the depths of 20 m, 40 m,
717 and 60 m was estimated to be ~ 0.1 MPa, ~ 0.22 MPa, and ~ 0.38 MPa, respectively. Note that the
718 slope of the surface of ice sheets and glaciers at Summit is idealized to be zero, i.e., their surfaces
719 are horizontal. The applied stress, σ , is the applied load divided by the cross-sectional area of a
720 sample. The σ at the depths of 20 m, 40 m, and 60 m were 0.21 MPa, 0.32 MPa, and 0.43 MPa,
721 respectively. The effective stress, $\tilde{\sigma}$, is defined as σ divided by the fraction of ice matrix in
722 firm, see in detail from Li and Baker (2022a). Thereby, $\tilde{\sigma}$ is 0.32 MPa (the mean porosity of
723 34.9%), 0.43 MPa (24.8%), and 0.5 MPa (14.4%) from the 20–60 m samples, respectively. Note
724 that the stresses were vertically loaded on the sample (parallel to the direction of core axis of the
725 sample) in laboratory tests. Ideally, in order to be analogous to the densification of firm in nature,
726 $\tilde{\sigma}$ for laboratory samples from a given depth should be equal to the p of firm *in situ* at an
727 equivalently same depth at Summit, namely $\tilde{\sigma}/p = 1$. However, in consideration of the laboratory
728 timeframe for experiments (Pimienta and Duval, 1987), the stresses applied in laboratory tests are
729 usually higher with a resulting higher rate of deformation than those *in situ*. Thus, to observe the
730 effect of the stress on the creep of firm with different densities at different depths, we designed the
731 following configuration of the $\tilde{\sigma}/p$ with depth, viz., $0.32 \text{ MPa}/\sim 0.1 \text{ MPa} = \sim 3.2$, $0.43 \text{ MPa}/\sim 0.22$
732 $\text{MPa} = \sim 1.95$, and $0.5 \text{ MPa}/\sim 0.38 \text{ MPa} = \sim 1.32$ for the samples from the depths of 20 m, 40 m,

733 and 60 m, respectively. In this manner, the decrement of $\tilde{\sigma}/p$ with increasing depth represents
734 the decrease of the effective stress with increasing depth. Also, it's important to note that the
735 strain rates achieved during creep experiments in laboratory settings are 6 to 7 times faster than
736 on ice sheets due to the constraints of conducting experiment in reasonable times, which requires
737 higher loads.

738

739 **Appendix B:** Strain rate minimum inferred via two kinds of constraints

740 To improve the reliability of inferred SRmins, two kinds of constraints were applied. First, the
741 SRmins from the -5°C and -18°C samples are calibrated using Glen's law $\dot{\epsilon}=A\sigma^n$ with $n = 4.3$
742 (Li and Baker, 2022a). PC1-SRmin, PC2-SRmin, and PC3-SRmin indicate three possibilities of
743 the SRmins that are calculated from the 20 m, 40 m, and 60 m samples via the *only* SRmin
744 observed at a given temperature (Table 2). As an example, for the -5°C samples, there exist three
745 possibilities from three depths. 1) The SRmin observed from the 20 m sample in bold italic font is
746 used to calculate two other SRmins for the 40 m and 60 m samples in the italic font in the column
747 of PC1-SRmin. 2) In the same manner as in scenario 1), the SRmin observed from the 40 m
748 sample is calculated in the column of PC2-SRmin in the bold italic font, and the SRmin observed
749 from the 60 m sample is calculated in the column of PC3-SRmin in the bold italic font. 3) In the
750 same manner as in scenarios 1) and 2), the SRmin is calculated for the -18°C samples in turn
751 from three depths. Second, the SRMin of the -30°C samples is inferred on the basis of the range
752 of Q_c , i.e., from 44.8 kJ mol^{-1} (upper bound) to 113 kJ mol^{-1} (lower bound), using the Arrhenius
753 relation.

754 **Data availability**

755 The data supporting the conclusions in this study are available at <https://arcticdata.io/catalog>.

756

757 **Author contribution**

758 Y.L. and I.B. designed the experiments and Y.L. carried them out. Y.L. analyzed the data and
759 visualized the relevant results. Y.L. prepared the manuscript with contributions from all
760 co-authors (K.K. and I.B.).

761

762 **Competing interests**

763 At least one of the (co-)authors is a member of the editorial board of The Cryosphere.

764

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