

# Luminescence and a New Approach for Detecting Heat Treatment of Geuda Sapphire

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**Abstract.** For decades, unravelling heat treatment of sapphire has been a challenging issue. The present study offers new aspects that support the detection of heat treatment of sapphire. Natural geuda sapphire exhibits orange to red luminescence under long-wave ultraviolet (LWUV, 365 nm) light, while heated geuda sapphire shows blue luminescence under short-wave ultraviolet (SWUV, 225 nm) light. UV-excited photoluminescence reveals a connection between the broad emission spectrum of unheated sapphire, which appears orange to red under LWUV illumination, and the emission spectrum of heated sapphire, which appears blue under SWUV illumination. The presence of melt inclusions in dissolved silks serves as an indicator of sapphire heat treatment. Fourier-transform infrared (FTIR) spectroscopy alone is insufficient for distinguishing unheated from heated sapphire. By combining orange to red luminescence with blue luminescence and melt inclusions, we provide a practical method for accurately differentiating natural gem-quality sapphire and heated gem-quality sapphire.

Keywords: gem; sapphire; heat treatment; luminescence

## 1 Introduction

Since the 1970s, Sri Lanka has been renowned for its gemstone resources, particularly geuda sapphire, a milky or silky corundum variety that is frequently heat-treated to enhance color and clarity (Soysa and Fernando, 1992; Ediriweera and Perera, 1989; Perera et al., 1991). High-temperature treatment of corundum (including ruby and sapphire) can significantly alter its milkiness, asterism, color, and internal

38 features such as mineral inclusions (Nassau, 1981; Ediriweera and Perera, 1989; Hughes, 1997, 2017; Kyi  
39 et al., 1999; Pisutha-Arnond, 2017; Themelis, 2018). Key factors in these transformations include the  
40 temperature, duration, and atmospheric conditions of the heating process (Nassau, 1981; Emmett and  
41 Douthit, 1993; Peiris, 1993; Emmett et al., 2003; Hughes, 2017; Pisutha-Arnond, 2017; Soonthorntantikul  
42 et al., 2019).

43 One of the first-rank challenges encountered by gemologists nowadays is the precise and  
44 reliable identification of heat-treated ruby and sapphire. Blue luminescence under SWUV light, observed  
45 in heated sapphire for over 50 years (Crowningshield 1966, 1970), can extend into the green region  
46 (Nassau, 1981) and has been extensively studied (Evans, 1994; Wong et al., 1995a; Wong et al., 1995b;  
47 Hughes 1997; McClure and Smith, 2000; Page et al., 2010; Alombert-Goget et al., 2016a; Alombert-Goget  
48 et al., 2016b; Hughes, 2017; Vigier et al., 2021a, b, 2023). This luminescence may relate to rutile inclusions  
49 commonly found in natural blue sapphire (Hughes, 2017). During heating at around 1600 °C, rutile  
50 decomposes, incorporating Ti<sup>4+</sup> ions into the sapphire structure (Sutthirat et al., 2006). Blue luminescence  
51 serves as a key indicator of heat treatment in sapphire (Crowningshield, 1966; McClure and Smith, 2000;  
52 Hughes, 2017; Hughes and Perkins, 2019). However, changes in original brown silk inclusions and orange  
53 luminescence in natural unheated sapphire, relative to blue luminescence in heated sapphire, have not been  
54 thoroughly explored. The present study provides a novel approach by observing the transformations in silk  
55 inclusions and luminescence in sapphire before and after heating

56

## 57 **2 Materials and Methods**

58

59 Natural unheated geuda sapphire (A-type) samples (as described by Vertriest et al., 2019) were separated  
60 based on the appearance of silk inclusions into three distinctive groups, i. e., high-density-silk (HS), low-  
61 density-silk (LS), and silk-free (SF) specimens (Fig. 1). Using an Enraf-Nonius Kappa single-crystal X-ray  
62 diffractometer (sXRD) with a charge-coupled device (CCD) area detector, these samples were oriented  
63 (based on 10 frames at a crystal detector distance of 35 mm), cut and polished into wafers with surfaces  
64 parallel to the *c*-axis. For chemical analysis using an electron probe micro-analyser (EPMA) system, slabs  
65 were coated with carbon for conductivity.

66 Heating experiments were conducted using a high-temperature electric furnace, Linn-  
67 HT-1800-Vac. Heating was performed under ambient atmospheric conditions without any additional  
68 oxygen buffer. Experimental conditions involved the maximum temperature of 1650 °C, which was  
69 maintained for 10 hours, prior to natural cooling down in the furnace. A heating rate of 300 °C per hour  
70 was set to reach the maximum temperature. To minimize surface contamination, the samples were placed  
71 into a highly purified alumina (Al<sub>2</sub>O<sub>3</sub>) crucible.

72 The refractive index of samples was measured by a gemological refractometer (KRUSS,  
73 model ER605) with 1.81 refractive index liquid. Specific gravity was determined by a hydrostatic weighing  
74 balance, by weighing samples in water (with a drop of dishwashing detergent added to reduce surface  
75 tension) and in air.

76 Micro-inclusions in all samples were imaged using an Olympus BX-series microscope  
77 equipped with Olympus DP27 digital camera. The camera was operated using the Olympus Stream micro-  
78 imaging software. Raman spectra of inclusions were acquired using a confocal micro-Raman spectrometer  
79 Horiba Jobin Yvon LabRAM-HR Evolution. Using 473 nm laser excitation (15 mW at the sample) and a  
80 50×/0.50 objective lens, a spectral range of 100–1350 cm<sup>-1</sup> Raman shift was recorded. Wavenumber  
81 calibration was done using the Rayleigh line, resulting in wavenumber accuracy of better than 0.5 cm<sup>-1</sup>. A  
82 spectral resolution of ca. 1.2 cm<sup>-1</sup> resulted from 800 mm focal length and an 1800 grooves/mm optical  
83 grating in the monochromator system. For more details see Zeug et al. (2018).

84 Chemical compositions of the samples were determined using a JEOL JXA 8100 EPMA.  
85 Analytical conditions were set to 15 kV accelerating voltage and a probe current of about  $2.5 \times 10^{-8}$  A with  
86 electron beam focussed to <1 μm. Natural mineral and synthetic oxide references were selected suitably  
87 for calibration, including fayalite (Fe<sub>2</sub>SiO<sub>4</sub>) for Fe, wollastonite (CaSiO<sub>3</sub>) for Ca, synthetic corundum  
88 (Al<sub>2</sub>O<sub>3</sub>) for Al, synthetic periclase (MgO) for Mg, synthetic quartz (SiO<sub>2</sub>) for Si, potassium titanyl  
89 phosphate (KTiOPO<sub>4</sub>) for K and Ti, synthetic manganosite (MnO) for Mn, synthetic eskolaite (Cr<sub>2</sub>O<sub>3</sub>) for  
90 Cr, synthetic gadolinium gallium garnet (Gd<sub>3</sub>Ga<sub>5</sub>O<sub>12</sub>) for Ga, and synthetic lead vanadium germanium  
91 oxide for V. Counting times were 600 s peak and 300 s background for all elements. The K-α line was  
92 analysed for all elements except for Ga where the L-α line was measured. Analytical crystals were selected  
93 appropriately including thallium acid phthalate (TAP) crystal for Si and Al; pentaerythriol (PET) crystal  
94 for Ti, Mg, K, and Ca; lithium fluoride (LIF) crystal for V, Cr, Ga, Fe, and Mn. The detection limit  
95 (estimated from threefold background noise) is approximated at 0.005 wt% or 50 ppm. Three spots in each  
96 sample were analyzed.

97 Polarized optical absorption (UV-VIS-NIR) spectra of samples were recorded on  
98 double-sided polished crystal slabs in the spectral range of 35000–3500 cm<sup>-1</sup>, covering the near ultraviolet  
99 (UV), the visible (VIS) and the near infrared (NIR) ranges. The measurements were performed in the  
100 sample chamber of a Bruker Vertex 80 FTIR spectrometer at 2 mm measuring spot, using a calcite Glan-  
101 prism polarizer and appropriate combinations of light sources (Xe or W lamp), beam splitters (CaF<sub>2</sub>-  
102 VIS/UV or CaF<sub>2</sub>-NIR), and detectors (GaP, Si or InGaAs diodes) to cover the desired spectral range.  
103 Hence, each full spectrum was combined from three partial spectra: 1) 35000–18000 cm<sup>-1</sup> with 40 cm<sup>-1</sup>  
104 spectral resolution and averaged from 256 scans; 2) 18000–9500 cm<sup>-1</sup> with 20 cm<sup>-1</sup> resolution and 256  
105 scans; 3) 9500–3500 cm<sup>-1</sup> with 10 cm<sup>-1</sup> resolution and 128 scans.

106 Fourier-transform infrared spectra were acquired by means of a Bruker Tensor 27 FTIR  
107 spectrometer attached to a Bruker Hyperion microscope in the spectral range from 4000 cm<sup>-1</sup> to 1600 cm<sup>-1</sup>.  
108 A glowbar light source, a KBr beamsplitter, and a deuterated L-alanine doped triglycene sulphate  
109 (dLATGS; Tensor27) or Hg-Cd-telluride (MCT) detector (Hyperion) were employed. The spectral  
110 resolution was 4 cm<sup>-1</sup>, sample and reference spectra were averaged from 128 scans.

111 Luminescence phenomena were observed and photo-captured both before and after heat  
112 treatment. The images were obtained under LWUV illumination using ZEISS microscope model stemi 508  
113 with 0.63x magnification. The images were captured in a darkened room using CANON digital single lens  
114 reflex (DSLR) camera model EOS 80D (24.2 MP resolution), which was mounted on top of the microscope.

115 The SUPERFIRE UV (365 nm) mini flashlight model S11–H, 3W (max), DC 3.7 V, was held approx. 15  
116 cm above the samples. The camera settings involved an exposure time of 5 s, an exposure bias of 0 steps,  
117 and an ISO speed of 200. The aperture was adjusted to  $f/0$ , and the focal length of 0 nm. For SWUV  
118 illumination (approximately 225 nm), a DiamondView™ device was used. The parameter settings for  
119 DiamondView™ were established as follows: Integration duration: 2.83 s; minimum excitation status: Off.  
120 Power settings ranged from 50% to 80%, contingent upon the intensity of luminescence. A gain of 13.85  
121 dB is measured. The aperture was set to 80% and the field stop to 67%. Gamma was disabled.  
122 Photoluminescence (PL) spectra in the visible and near–infrared ranges were acquired using a confocal  
123 Horiba Jobin Yvon LabRAM–HR 800 spectrometer. Spectra were excited using the 325 nm emission of a  
124 He–Cd laser (ca. 10 mW at the sample surface). The system was calibrated using emission lines of a Kr  
125 lamp. The spectral resolution was in the range 0.07 nm (violet) to 0.02 nm (NIR range). All the spectra  
126 were acquired at the same position both before and after heating experiments.

127

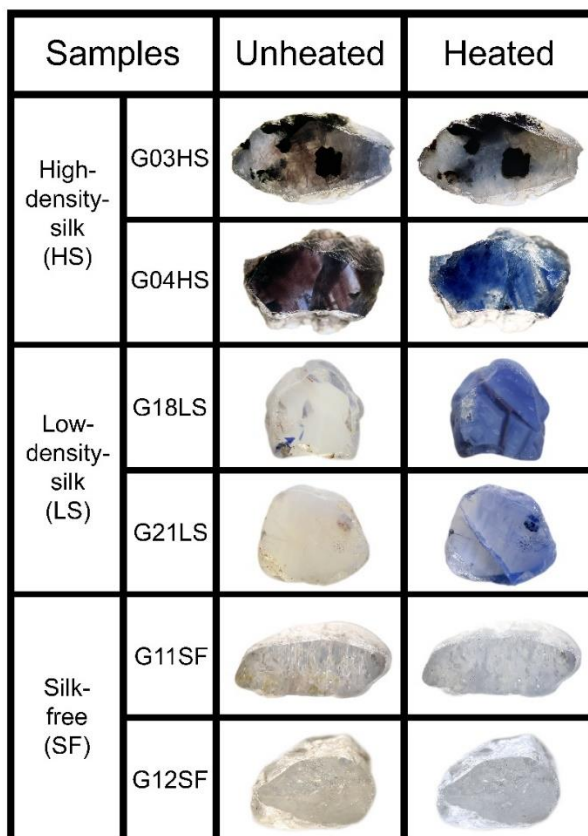
### 128 **3 Results**

129

#### 130 **3.1 Heating–induced property changes and alteration**

131 Representatives of natural unheated and their heated counterparts of all groups are shown in Fig. 1. All the  
132 samples ranged from a specific gravity of 3.83 to 4.08, and refractive indices of 1.760 to 1.770 falling well  
133 within the range of corundum properties. Before heating, samples showed varying natural appearances  
134 based on inclusion density. Geuda samples with HS inclusions (e.g., G03HS and G04HS, Fig. 1) exhibited  
135 brown silk and brown color banding or zoning, while a few samples also displayed a natural blue color.  
136 Samples with LS inclusions (e.g., G18LS and G21LS, Fig. 1) generally appeared milky with yellowish or  
137 brownish tints. After heating, most samples turned blue, ranging from pale to dark shades, with the milky  
138 appearance and yellowish or brownish tints significantly reduced. On the other hand, the SF group usually  
139 showed a slightly yellowish appearance (Fig. 1, samples G11SF and G12SF). After the heating experiment,  
140 they had changed slightly to a very pale blue color.

141



142

143 **Figure 1.** Representatives of natural unheated geuda sapphire samples within three separate groups, i.e.,  
 144 HS (G03HS, G04HS), LS (G18LS, G21LS), and SF (G11SF, G12SF) groups, and their appearances after  
 145 heating. Sizes of stones range between 4 and 12 mm.

146

### 147 3.2 Mineral chemistry

148 Chemical compositions of samples in the three distinct groups are summarized in Tables 1 to 3. The  $\text{Al}_2\text{O}_3$   
 149 contents range between 98.2 and 99.5 wt%. Other elements are found as trace contents only, particularly  
 150 Fe, Ti, and Ga. Fe and Ti are essential coloring elements in sapphire. The HS group contained the highest  
 151 Fe contents of 0.32–0.36 wt% FeO, together with 0.02–0.04 wt%  $\text{TiO}_2$  and <0.7 wt%  $\text{Ga}_2\text{O}_3$ . The LS group  
 152 had a high Ti content of 0.02–0.51 wt%  $\text{TiO}_2$  with  $\leq 0.06$  wt% FeO and  $\leq 0.8$  wt%  $\text{Ga}_2\text{O}_3$ . The SF group  
 153 contained 0.06–0.26 wt% FeO,  $\leq 0.04$  wt%  $\text{TiO}_2$  and <1 wt%  $\text{Ga}_2\text{O}_3$ .

154

155 **Table 1.** Representative chemical compositions (EPMA results) and calculated mineral formulae of HS  
 156 sapphire samples.

Samples	G01HS	G02HS	G03HS	G04HS
Major oxides (wt%):				
SiO <sub>2</sub>	0.00	0.00	0.45	0.40
TiO <sub>2</sub>	0.02	0.02	0.04	0.03
Al <sub>2</sub> O <sub>3</sub>	99.0	98.7	98.9	98.7
V <sub>2</sub> O <sub>3</sub>	0.01	0.00	0.00	0.03
Cr <sub>2</sub> O <sub>3</sub>	0.00	0.02	0.03	0.00
Ga <sub>2</sub> O <sub>3</sub>	0.62	0.66	0.00	0.39
FeO <sub>total</sub> *	0.32	0.33	0.36	0.36
MnO	0.02	0.00	0.00	0.02
MgO	0.00	0.00	0.01	0.01
K <sub>2</sub> O	0.00	0.00	0.00	0.00
CaO	0.01	0.01	0.02	0.01
Total	100.0	99.8	99.8	100.0
Mineral formulae (apfu)**:				
Si	0.000	0.000	0.008	0.007
Ti	0.000	0.000	0.001	0.000
Al	1.990	1.989	1.985	1.982
V	0.000	0.000	0.000	0.000
Cr	0.000	0.001	0.000	0.000
Ga	0.007	0.007	0.000	0.004
Fe	0.013	0.014	0.005	0.005
Mn	0.000	0.000	0.000	0.000
Mg	0.000	0.000	0.000	0.000
K	0.000	0.000	0.000	0.000
Ca	0.000	0.000	0.000	0.000
Sum	2.010	2.011	1.999	2.000

157 \* FeO<sub>total</sub> = total Fe oxide, assuming all Fe to be ferrous

158 \*\* Calculated based on 3 O atoms per formula unit

159

160 **Table 2.** Representative chemical compositions (EPMA results) and calculated mineral formulae of LS  
 161 sapphire samples.

Samples	G06LS	G16LS	G18LS	G20LS	G21LS
Major oxides (wt%):					
SiO <sub>2</sub>	0.00	0.00	0.11	0.00	0.00
TiO <sub>2</sub>	0.04	0.27	0.37	0.51	0.02
Al <sub>2</sub> O <sub>3</sub>	99.0	98.5	98.4	98.3	98.6
V <sub>2</sub> O <sub>3</sub>	0.02	0.03	0.01	0.02	0.01
Cr <sub>2</sub> O <sub>3</sub>	0.00	0.06	0.00	0.02	0.00
Ga <sub>2</sub> O <sub>3</sub>	0.10	0.37	0.58	0.81	0.59
FeO <sub>total</sub> *	0.00	0.06	0.06	0.06	0.05
MnO	0.01	0.00	0.00	0.00	0.01
MgO	0.02	0.02	0.01	0.01	0.02
K <sub>2</sub> O	0.00	0.00	0.00	0.00	0.00
CaO	0.00	0.00	0.01	0.01	0.00
Total	99.2	99.3	99.6	99.7	99.3
Mineral formulae (apfu)**:					
Si	0.000	0.000	0.002	0.000	0.000
Ti	0.001	0.003	0.005	0.007	0.000
Al	1.997	1.989	1.984	1.981	1.992
V	0.000	0.000	0.000	0.000	0.000
Cr	0.000	0.002	0.000	0.001	0.000
Ga	0.001	0.004	0.006	0.009	0.007
Fe	0.000	0.002	0.002	0.002	0.002
Mn	0.000	0.000	0.000	0.000	0.000
Mg	0.001	0.001	0.001	0.000	0.001
K	0.000	0.000	0.000	0.000	0.000
Ca	0.000	0.000	0.000	0.000	0.000
Sum	2.001	2.003	2.000	2.000	2.002

162 \* FeO<sub>total</sub> = total Fe oxide, assuming all Fe to be ferrous

163 \*\* Calculated based on 3 O atoms per formula unit

164

165 **Table 3.** Representative chemical compositions (EPMA results) and calculated mineral formulae of SF  
 166 sapphire samples.

Samples	G07SF	G11SF	G12SF	G14SF	G22SF	G23SF
Major oxides (wt%):						
SiO <sub>2</sub>	0.01	0.13	0.06	0.00	0.00	0.00
TiO <sub>2</sub>	0.03	0.04	0.03	0.00	0.04	0.01
Al <sub>2</sub> O <sub>3</sub>	98.7	98.7	98.8	99.5	98.2	98.7
V <sub>2</sub> O <sub>3</sub>	0.00	0.02	0.00	0.00	0.00	0.00
Cr <sub>2</sub> O <sub>3</sub>	0.00	0.00	0.01	0.00	0.00	0.00
Ga <sub>2</sub> O <sub>3</sub>	0.71	0.00	0.00	0.15	0.78	0.94
FeO <sub>total</sub> *	0.06	0.26	0.08	0.10	0.22	0.13
MnO	0.01	0.02	0.00	0.01	0.02	0.00
MgO	0.00	0.02	0.00	0.00	0.00	0.00
K <sub>2</sub> O	0.00	0.01	0.00	0.00	0.00	0.00
CaO	0.02	0.00	0.02	0.01	0.01	0.01
Total	99.6	99.2	99.0	99.8	99.3	99.8
Mineral formulae (apfu)**:						
Si	0.000	0.002	0.001	0.000	0.000	0.000
Ti	0.000	0.001	0.000	0.000	0.001	0.000
Al	1.991	1.993	1.997	1.997	1.988	1.988
V	0.000	0.000	0.000	0.000	0.000	0.000
Cr	0.000	0.000	0.001	0.000	0.000	0.000
Ga	0.008	0.000	0.000	0.002	0.009	0.010
Fe	0.002	0.011	0.003	0.004	0.009	0.005
Mn	0.000	0.000	0.000	0.000	0.000	0.000
Mg	0.000	0.001	0.000	0.000	0.000	0.000
K	0.000	0.000	0.000	0.000	0.000	0.000
Ca	0.000	0.000	0.000	0.000	0.000	0.000
Sum	2.002	2.009	2.003	2.003	2.007	2.004

167 \* FeO<sub>total</sub> = total Fe oxide, assuming all Fe to be ferrous

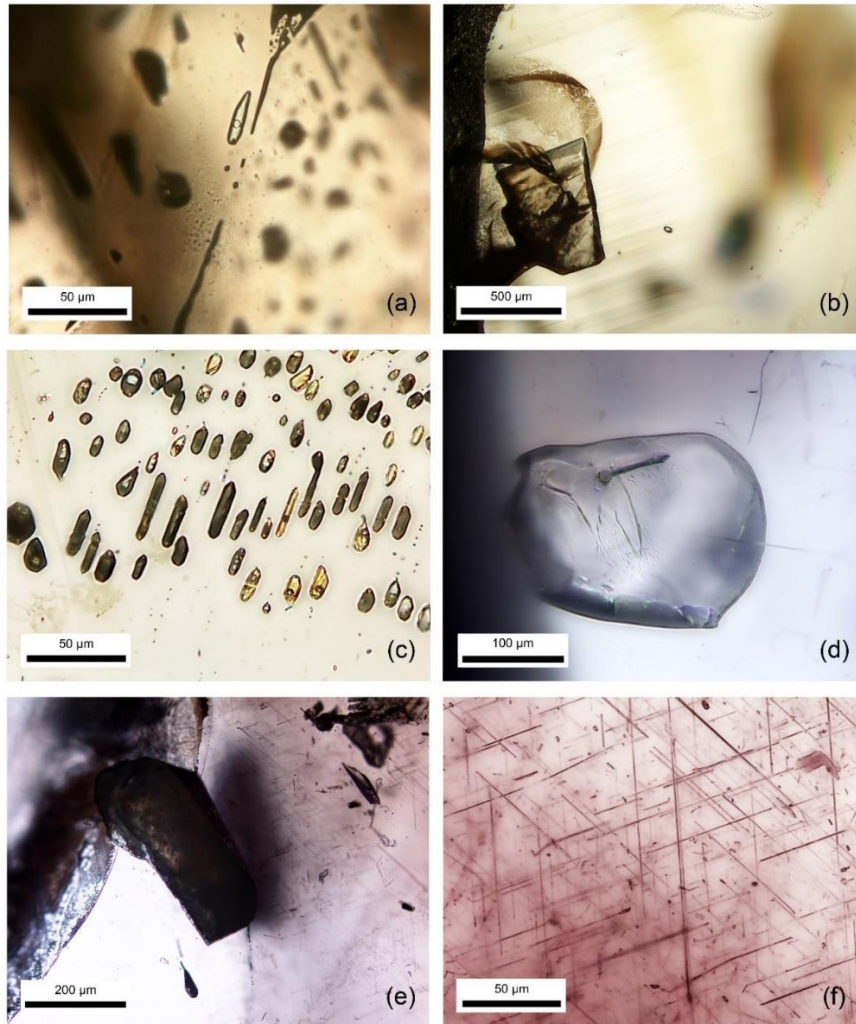
168 \*\* Calculated based on 3 O atoms per formula unit

169



### 170 3.3 Microscopic features

171 Negative crystals, with or without CO<sub>2</sub> gas bubbles, were commonly observed alongside mineral inclusions,  
 172 such as oligoclase feldspar, calcite, and muscovite, in these sapphire samples. Brown silk inclusions were  
 173 prominent in both HS and LS groups, as shown in Fig. 2 (G22SF, G18LS, and G04HS for Fig. 2a, Fig. 2b–  
 174 c, and Fig. 2d–f, respectively). Micro-Raman spectroscopy was used to identify CO<sub>2</sub> and mineral  
 175 inclusions. Although brown silk inclusions, typically needle-shaped and aligned with color banding (Fig.  
 176 3a), were often less than 1 μm in diameter and difficult to identify, irregular or flaky platelet forms (Fig.  
 177 3c) were also noted. 3c) were also noted.

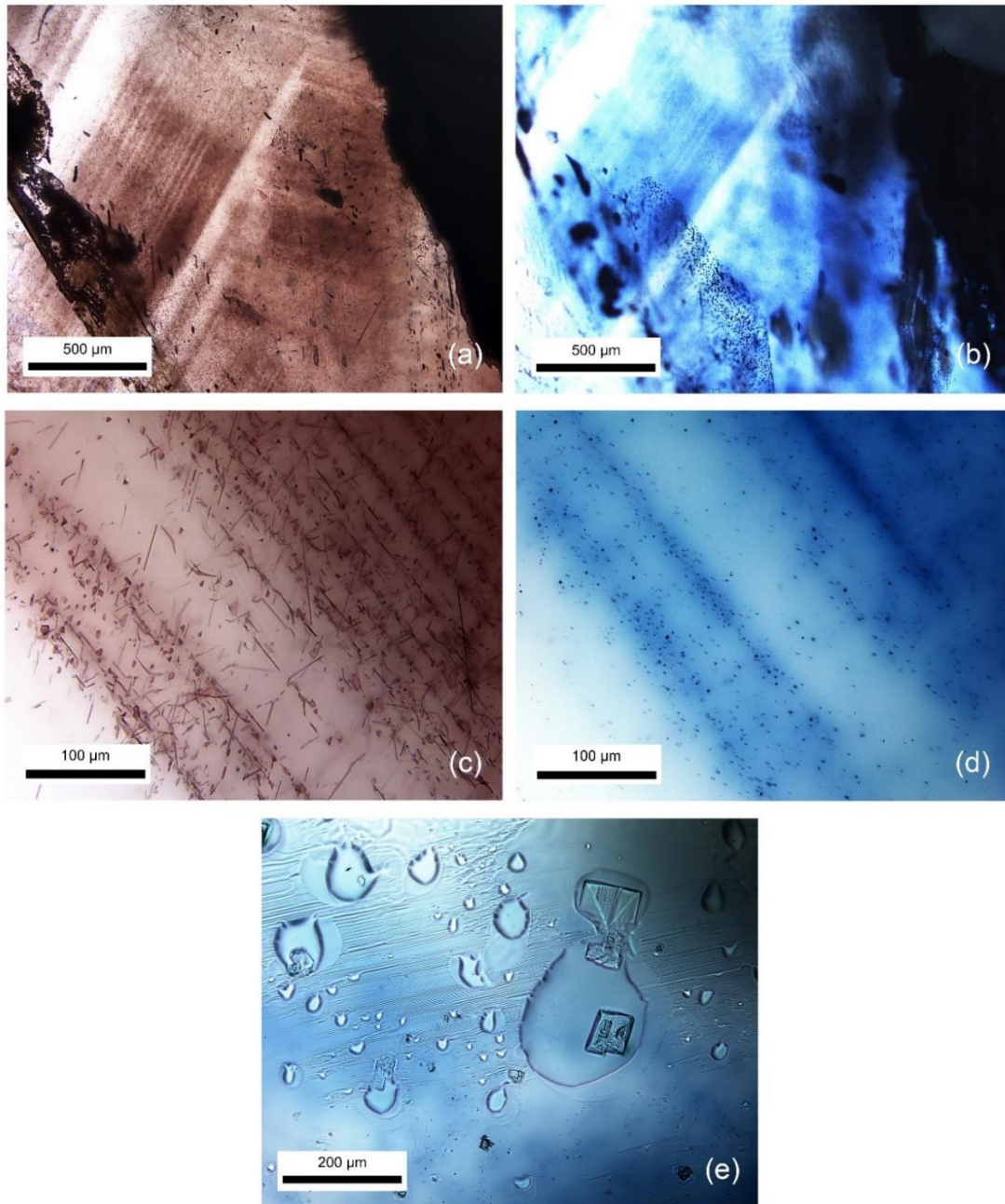


178  
 179 **Figure 2.** Transmitted-light photomicrographs of inclusions including CO<sub>2</sub>-containing negative crystals  
 180 (a), calcite (b), cluster of negative crystals (c), oligoclase (d), muscovite (e), and brown silks (f) in natural  
 181 unheated sapphire.

182

183 After high-temperature heating, molten surfaces (Fig. 3e) and decomposed crystal  
 184 inclusions were commonly observed in these samples. The most notable alteration was also detected in the  
 185 initial area of brown silks (Fig. 3a), which exhibited distinct bluish color banding/zoning (Fig. 3b) after  
 186 heating.

187



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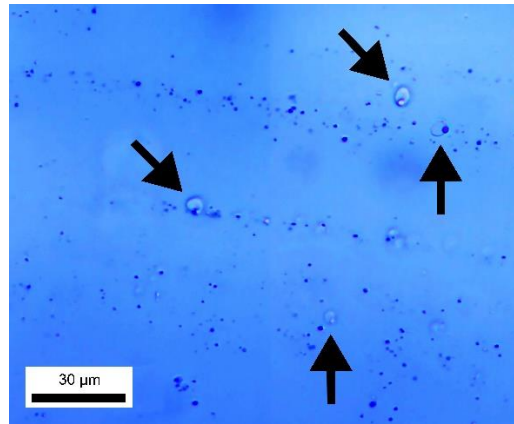
189 **Figure 3.** Transmitted-light photomicrographs showing that brown banding (a) with irregular platy  
 190 brownish flakes and tiny needles (c) in natural unheated sapphire sample G04HS turned into blue  
 191 color banding (b) with blue dots (d) upon heating. Melted surface (e) was also observed after heat treatment.

192

193

194 The brown silks (Fig. 3c) experienced a transformation upon heating into blue dots (Fig.  
 195 3d). Additionally, melt inclusions among blue dots were likely developed by melting of brown silks with  
 196 collaborative reaction of the sapphire host, which have never been reported elsewhere, becoming  
 197 significantly noticeable and useful for indicating heat treatment of sapphire (Fig. 4).

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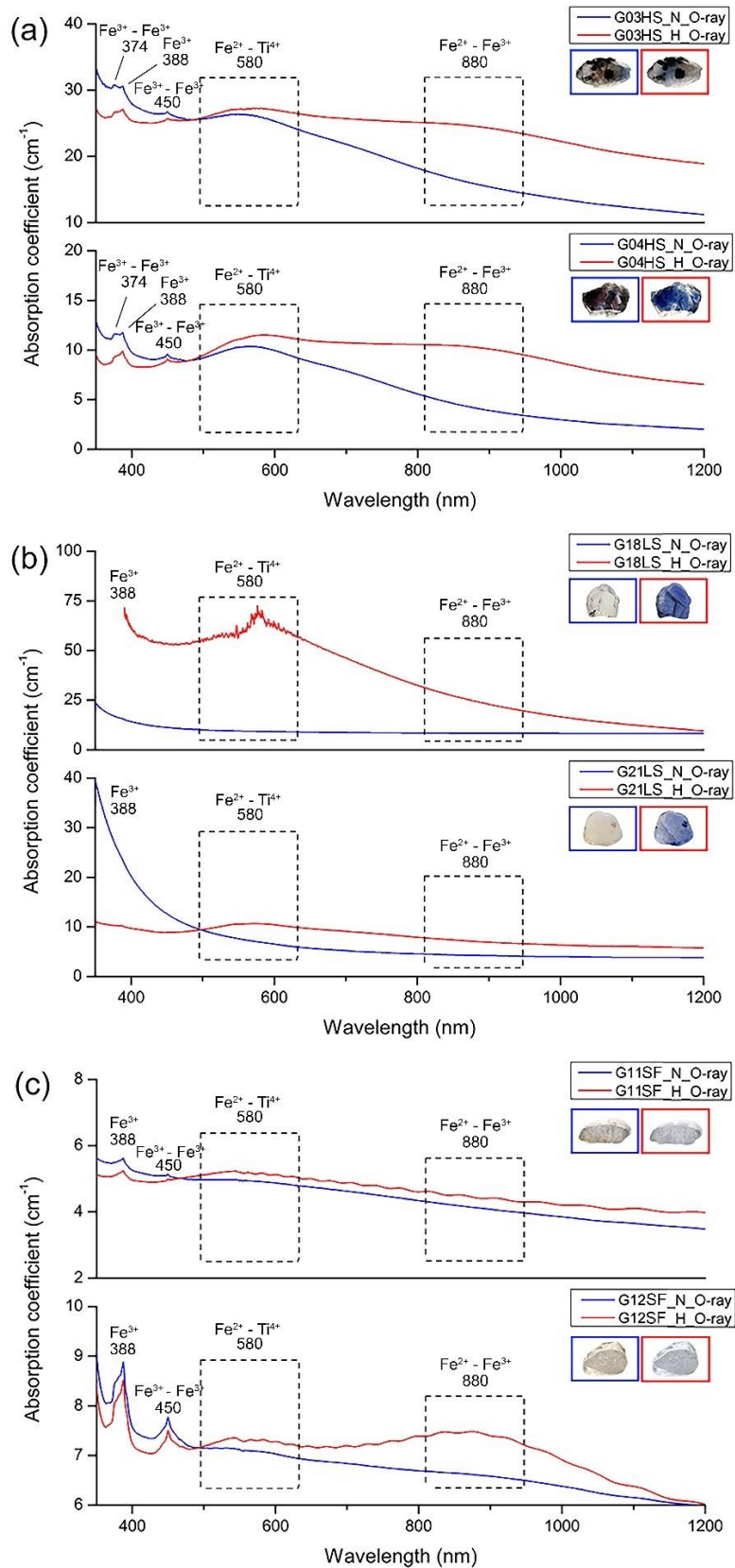
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199 **Figure 4.** Transmitted-light photomicrograph showing melt inclusions (arrows) among blue dots  
200 transformed from silk inclusions in sapphire after heating (sample G04HS).

201

### 202 **3.4 Optical (UV-VIS-NIR) spectroscopy**

203 The optical spectra of representative sapphire samples are presented in Fig. 5. Absorption peaks at 374,  
204 388, and 450 nm, as well as bands around 580 and 880 nm, were observed. Optical spectra have been  
205 studied on unheated and heated sapphire by numerous previous researchers (e. g., Ediriweera and Perera,  
206 1989; Perera et al., 1991; Emmett and Douthit, 1993; Hughes, 1997; Kyi et al., 1999; Emmett et al., 2003;  
207 Sripoonjan et al., 2014; Hughes, 2017; Pisutha-Arnond, 2017; Themelis, 2018; Palke et al., 2019;  
208 Soonthorntantikul et al., 2019; Dubinsky et al., 2020). The 374, 388, and 450 nm peaks as well as the 880  
209 nm band were proposed to be attributed to Fe, the 580 nm band to the Fe-Ti pair. After heating, all samples  
210 showed a significant increase in the main Fe-Ti pair related absorption band at around 580 nm (Figs. 5a-  
211 c), whereas Fe-Fe related absorption at around 880 nm was obviously increased in some samples (i. e.,  
212 Figs. 5a and 5c). The intensified absorption of the 580 nm band in these samples is referred to an increase  
213 of Fe-Ti pairs after heating which leads to enhanced blue coloration in heated sapphires.



214

215 **Figure 5.** Optical absorption spectra of untreated (blue lines) and heated (red lines) samples: (a) HS group

216 (G03HS, G04HS); (b) LS group (G18LS, G21LS); (c) SF group (G11SF, G12SF). Sizes of stones range

217 between 4 and 12 mm.

218

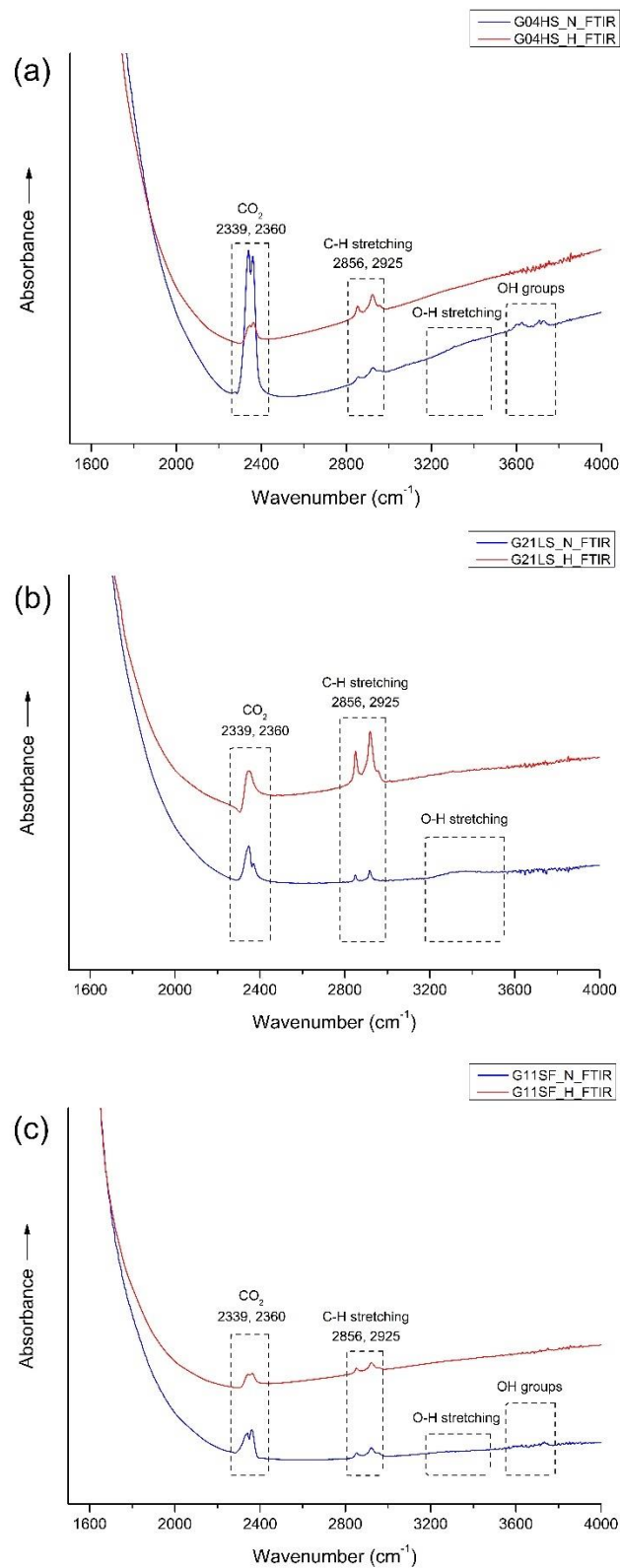
219 Spectral characteristics of corundum containing  $\text{Fe}^{3+}$  ions exhibit a high degree of  
 220 complexity. It is noteworthy that  $\text{Fe}^{3+}$  has electron configuration  $d^5$  resulting in a crystal field spectrum  
 221 with ground state  ${}^6\text{A}_1$  (Ferguson and Fielding, 1972). Small peaks at 374 nm ( ${}^4\text{E}^b$ ) and 450 nm ( ${}^4\text{A}_1, {}^4\text{E}^a$ )  
 222 should be attributed to the enhanced absorption of  $\text{Fe}^{3+}$ – $\text{Fe}^{3+}$  pairs (McClure, 1962; Ferguson and Fielding,  
 223 1971; Krebs and Maisch, 1971; Ferguson and Fielding, 1972) as well as a weak broadband absorption at  
 224 540 nm ( ${}^4\text{T}_2$ ) which could not be seen in this work. The distinct peak observed at a wavelength of 388 nm  
 225 ( ${}^4\text{T}_2^b$ ) (Krebs and Maisch, 1971) is linked to the individual  $\text{Fe}^{3+}$  ions. This, however, does not rule out the  
 226 possibility of a higher-order cluster with extra ions or other point defects (Emmett et al., 2003).  
 227 Additionally, there is also a broad band at a wavelength of 330 nm ( ${}^4\text{T}_1^b$ ) which is interpreted as a  $\text{Fe}^{3+}$ –  
 228  $\text{Fe}^{3+}$  pair absorption (Ferguson and Fielding, 1972). This is also present in the spectra of heated samples  
 229 G03HS and G04HS, as well as in all spectra of sample G12SF in this study. In trace contents both  $\text{Fe}^{2+}$  ( $d^6$ )  
 230 and  $\text{Ti}^{4+}$  ( $d^0$ ) ions alone do not exhibit any absorption in corundum in the visible range (Townsend, 1968);  
 231 on the other hand,  $\text{Fe}^{2+}$ – $\text{Ti}^{4+}$  pairs ( $t_2 \rightarrow {}^2\text{E}$ ) (Ferguson and Fielding, 1971) may yield a broad band  
 232 absorption around 580 nm (E $\perp$ c), or 700 nm (E $\parallel$ c) (Dubinsky et al., 2020). The  $\text{Fe}^{2+}$ – $\text{Fe}^{3+}$  pair gives rise to  
 233 the broad absorption band at ca. 880 nm (Fig. 5; Ferguson and Fielding, 1972).

234

### 235 3.5 FTIR spectroscopy

236 FTIR spectra of most samples yielded identical patterns within the range of 1600–4000  $\text{cm}^{-1}$  (Fig. 6). They  
 237 usually showed  $\text{CO}_2$  peaks (at 2339 and 2360  $\text{cm}^{-1}$ ), as well as C–H stretching related peaks (at 2856 and  
 238 2925  $\text{cm}^{-1}$ ), likely from artefacts (Fig. 6, blue lines), in accordance with Hughes (2017) and  
 239 Soonthorntantikul et al. (2021). However, O–H stretching of boehmite/diaspore peaks (at 1975 and 2105  
 240  $\text{cm}^{-1}$ ) (Delattre et al., 2012; Sun et al., 2015; Choi et al., 2017; Filatova et al., 2021; Soonthorntantikul et  
 241 al., 2021) was only observed in sample G03HS (Fig. 7a, blue line). Weak absorption features of O–H  
 242 stretching from  $\text{H}_2\text{O}$  (broad band at ca. 3400  $\text{cm}^{-1}$ ) and OH groups (ca. 3600–3700  $\text{cm}^{-1}$ ) were only found  
 243 in the untreated samples (blue lines), see Fig. 6a.

244

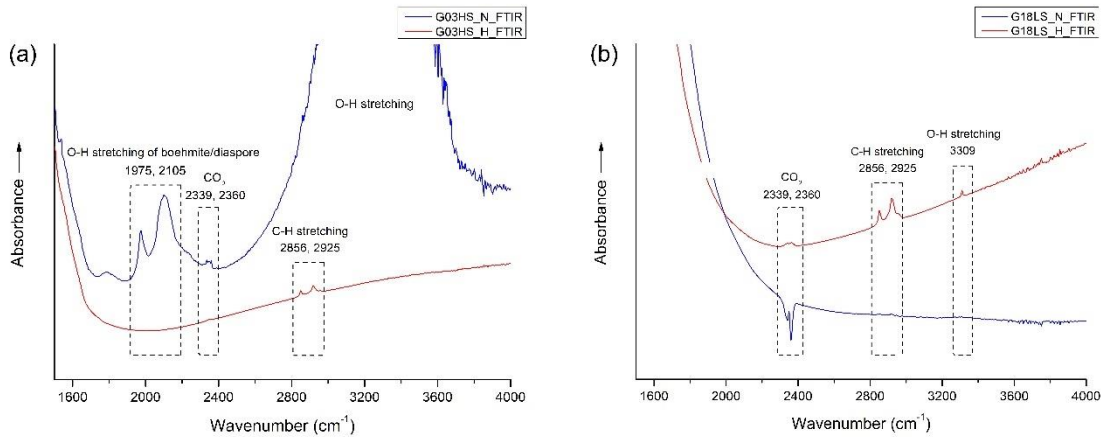


245

246 **Figure 6.** FTIR spectra obtained before (blue lines) and after (red lines) heating experiments of  
 247 representative samples G04HS (a), G21LS (b), and G11SF (c), respectively.

248

249 After heating, boehmite/diaspore-related absorption peaks (only observed in sample  
 250 G03HS, Fig. 7a) at 1975 and 2105  $\text{cm}^{-1}$  disappeared. In contrast, the 3309  $\text{cm}^{-1}$  hydroxyl (O-H) absorption,  
 251 which was not present in any natural sample before heating, appeared only in sample G18LS after heating  
 252 (Fig. 7b, red line).  
 253



254  
 255 **Figure 7.** FTIR spectra obtained before (blue lines) and after (red lines) heat treatment of samples G03HS  
 256 (a) and G18LS (b).  
 257

258

259 In this study, an absence of O–H absorption in the 3100–3600  $\text{cm}^{-1}$  range in all natural  
 260 geuda sapphire samples, together with the development of a weak absorption at 3309  $\text{cm}^{-1}$  upon heating in  
 261 only one of the samples (see Fig. 7b, red line), address the limitation to differentiate unheated and heated  
 262 sapphire by FTIR spectroscopy. Furthermore, heat treatment employed in this study did not involve the use  
 263 of any additional gases, such as hydrogen, to create a reducing atmosphere within the furnace. Despite this,  
 264 the 3309  $\text{cm}^{-1}$  absorption band was seen after the heating process. This might be in accordance with an  
 265 explanation proposed earlier by Notari et al. (2018).

266

267 The controversy of the presence of an O–H peak in the FTIR spectrum in unheated and  
 268 heated sapphire could be attributed to an inherent hydrogen content of the corundum. Hydrogen was found  
 269 in corundum, primarily in the form of alumina hydrates (Notari et al., 2018). These hydrates could release  
 270 hydrogen through de-hydroxylation at temperatures as low as approx. 450 °C. Additionally,  
 271 hydrogen was present in the air as H<sub>2</sub>O, which can be split at temperatures around 900 °C to produce  
 272 hydrogen gas (H<sub>2</sub>) and oxygen gas (O<sub>2</sub>) through the reaction  $2\text{H}_2\text{O} \rightarrow 2\text{H}_2 + \text{O}_2$  (Notari et al., 2018).

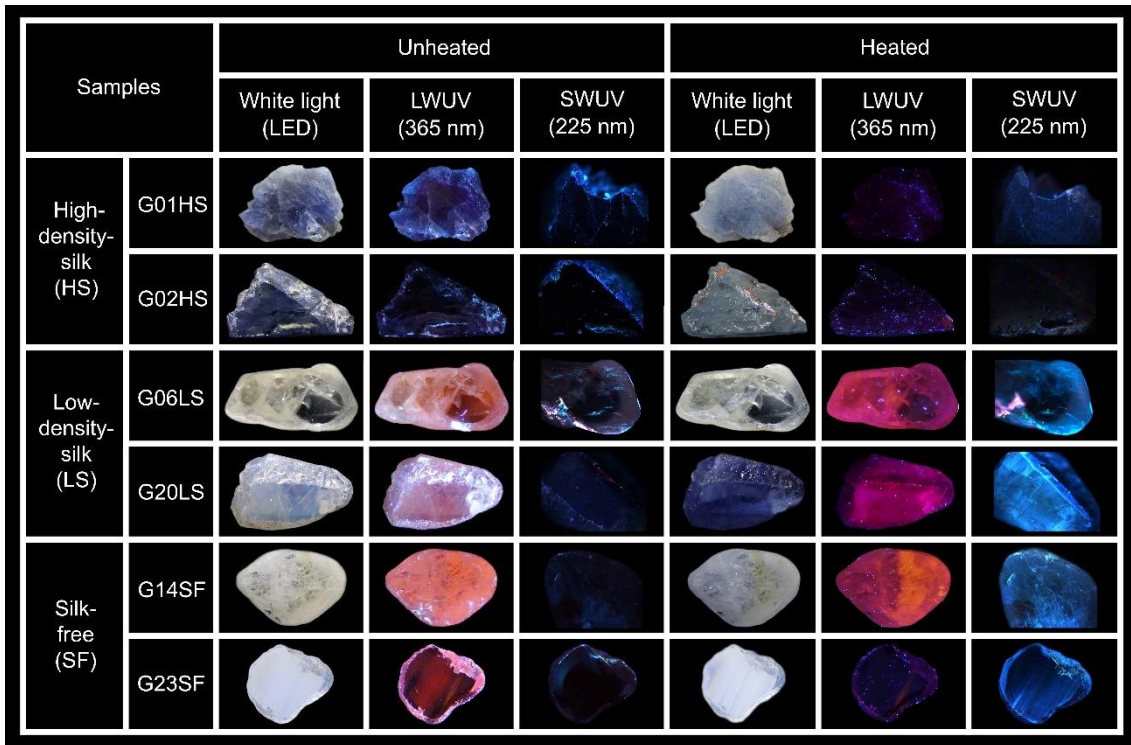
273

### 274 3.6 Photoluminescence imaging and spectroscopy

275

276 Photos presenting luminescence of some samples both before and after heat treatment are  
 277 shown in Fig. 8. Before heating, all natural sapphire samples were inert to SWUV light; moreover, all LS  
 278 and SF samples exhibited orange to red luminescence under LWUV light (Fig. 8). After heating, all LS and  
 279 SF samples, exhibited intense blue luminescence under SWUV light whereas their initial orange to red  
 280 luminescence under LWUV light turned into a strong purplish red luminescence (Fig. 8, samples G06LS  
 281 and G20LS in particular). In summary, the HS samples were all inert to SWUV and LWUV light both

279 before and after heating. Notably under LWUV light, an initial orange to red luminescence of a few samples  
 280 from the SF group was drastically reduced after heating (e.g., G23SF in Fig.8).  
 281



282  
 283 **Figure 8.** Representative images of HS (G01HS, G02HS), LS (G06LS, G20LS), and SF (G14SF, G23SF)  
 284 groups obtained under LWUV and SWUV illumination before and after heating. Sizes of stones range  
 285 between 4 mm and 12 mm.

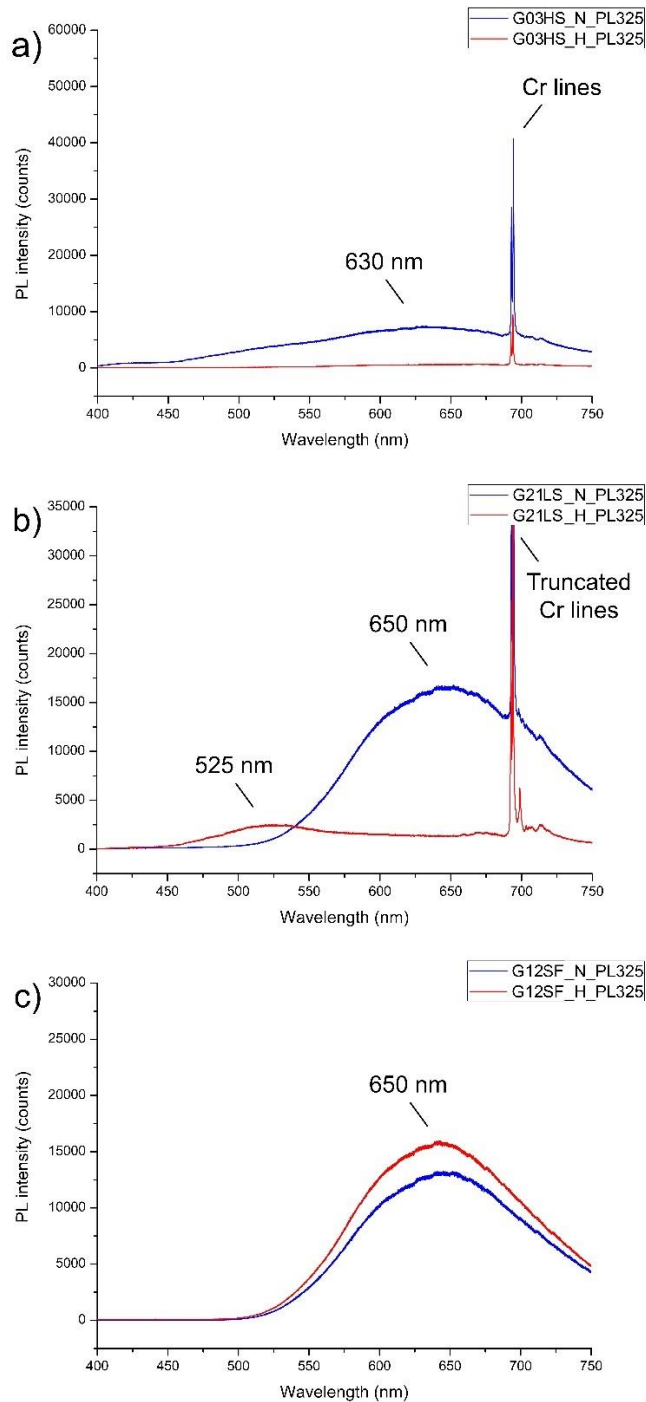
286  
 287 The UV-excited PL spectra showed that all the unheated and heated sapphire samples  
 288 have an identical feature of two narrow peaks of trace  $\text{Cr}^{3+}$  lines at around 692.8 and 694.2 nm (Fig. 9) that  
 289 are assigned to the spin-forbidden  ${}^2E \rightarrow {}^4A_2$  relaxation of trace  $\text{Cr}^{3+}$  (Nelson and Sturge, 1965). However,  
 290 the  $\text{Cr}^{3+}$  lines of some samples (Fig. 9c) are too weak to be visible within the noise of a broad and strong  
 291 emission band. All unheated sapphire samples showed a similar emission band in the orange to red region  
 292 centered around 630–650 nm (Fig. 9a–c, blue line). Remarkably, this appears to be associated with orange  
 293 to red luminescence under LWUV light, as noted by (Segura, 2013; Vigier et al., 2021a, b, c; Vigier and  
 294 Fritsch, 2022). Despite having the emission band around 630–650 nm, only unheated sapphire from HS  
 295 group appeared inert under LWUV illumination while the others revealed orange to red luminescence.

296 After heating, significant alteration in the emission band was observed, as depicted by  
 297 the red lines in Fig. 9a–c. The photoluminescence spectra of sample G03HS exhibited a notable reduction  
 298 in the emission band through the visible region (Fig. 9a, red line). This went along with a lack of  
 299 luminescence both under SWUV and LWUV excitation, whereas sample G12SF demonstrated a slight  
 300 increase of the emission band in the orange to red region (Fig. 9c, red line). More details are given in the  
 301 discussion part below.

302 In contrast to the other groups, after heating, sample G21LS (Fig. 9b, red line) exhibited  
 303 a significant emission band in the green region at around 525 nm. Note that this broad emission is excited



304 with the 325 nm laser (Fig. 9) but does not seem to affect significantly the emission colors observed under  
 305 SWUV (225 nm) and LWUV (365 nm) excitation (Fig. 8). For a discussion of the possibly strong  
 306 dependence of emission intensity (and color) on the excitation wavelength see for instance Zeug et al.  
 307 (2022). Likewise heated sapphire has been proposed to have an emission band in the blue region, which  
 308 corresponds to blue luminescence under SWUV light (Nassau, 1981; Hughes, 2017; Vigier et al., 2023).  
 309



310

311 **Figure 9.** Representative photoluminescence (UV-excited) spectra obtained before (blue lines) and after  
 312 (red lines) heating of sample G03HS (a), sample G21LS (b), and sample G12SF (c).

313

## 314 4 Discussion

315

### 316 4.1 Generalities

317 After  $\text{Ti}^{4+}$  being exposed to SWUV light, they yield luminescence (Nasdala and Fritsch,  
 318 2024). However, blue luminescence was not observed in both unheated and heated basaltic sapphire,  
 319 possibly due to the abundant presence of  $\text{Fe}^{2+}$  of basaltic origin that may strongly quench such blue  
 320 luminescence (Soonthorntantikul et al., 2019). More details will be discussed in this report. Furthermore,  
 321 even though microscopic inclusions have been the distinguishing characteristics of heated sapphire,  
 322 identifying heat-treated sapphire remains challenging (Crowningshield, 1966; Hughes, 2017). FTIR  
 323 spectroscopy has also been applied to detect heated sapphire. In some cases, the presence or absence of  
 324 specific FTIR features in the O–H absorption region ( $3100\text{--}3600\text{ cm}^{-1}$ ) may serve as an indicator of heat  
 325 treatment (Smith, 1995; Beran and Rossman, 2006; Saeseaw et al., 2018); however, it is probably not a  
 326 conclusive evidence (Ediriweera and Perera, 1989; Perera, 1993; Sutthirat et al., 2006; Cartier, 2009; Jaliya  
 327 et al., 2020). For example, the presence of the  $3309\text{ cm}^{-1}$  FTIR absorption peak was used as an indicator of  
 328 heated corundum (Hughes and Perkins, 2019; Soonthorntantikul et al., 2021). However, recent discoveries  
 329 show that this peak may also be found in unheated sapphire, suggesting that it is not a reliable indication  
 330 of heat treatment (Hughes, 1997, 2017; Hughes and Perkins, 2019; Soonthorntantikul et al., 2021).

331

### 332 4.2 Silk inclusions and coloration of sapphire

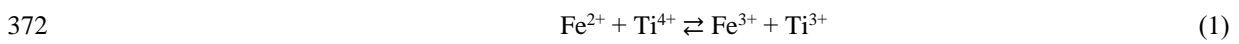
333 Studies addressing brown silk inclusions in corundum are scarce. Soonthorntantikul et al. (2021) reported  
 334 a mix of whitish silk and irregular/flaky/platelet-like brownish silk inclusions in corundum from Mogok.  
 335 Brown silk was ascribed as presumable ilmenite ( $\text{FeTiO}_3$ ) which is noticeable in high-Fe sapphire, whereas  
 336 white silk was suggested to consist of rutile ( $\text{TiO}_2$ ). The brown silks seen in our sapphire samples are likely  
 337 ilmenite, which is supported by their irregular/flaky/platelet-like brownish appearance and their high Fe  
 338 and Ti contents (note that the highest quantity of Fe was found in the HS group). Ilmenite decomposition  
 339 upon heat treatment does result in Fe and Ti migration into the host sapphire and subsequently causes blue  
 340 coloration. In particular, the decomposition of brown silks during heat treatment induces the formation of  
 341 blue dots, which is a result of the  $\text{Fe}^{2+}\text{--Ti}^{4+}$  pairing formation. Upon closer inspection using a high-  
 342 resolution microscope, these blue dots reveal distinct micro-inclusions as melt inclusions (with size of  $\leq 1$   
 343  $\mu\text{m}$ , see Fig. 4), which have never been documented before. However, it should be noted that these melt  
 344 inclusions are possibly derived from the decomposition of silks.

345 This work focuses only on blue coloration in sapphire which mainly relates to the  $\text{Fe}^{2+}\text{--}$   
 346  $\text{Ti}^{4+}$  pair, as initially noted by Townsend (1968), followed by Mattson and Rossman (1988), Moon and  
 347 Phillips (1994), and Emmett et al. (2003). Ti exhibits electron-donor properties, whereas Fe may function  
 348 as an electron acceptor. When occupying neighboring  $\text{Al}^{3+}$  positions, absorption due to intervalence charge  
 349 transfer between such donor-acceptor pairs may occur (details reported by Emmett et al., 2003 and  
 350 Monarumit et al., 2023).

351 It should also be mentioned that  $\text{Ti}^{4+}$  ions do not exhibit any absorption characteristics in  
 352 the visible spectrum when considered individually. The  $\text{Ti}^{4+}$  ion has a closed-shell electron configuration,

353 whereas the Fe<sup>2+</sup> ion mainly absorbs wavelengths within the near infrared and low-energy visible regions.  
 354 In contrast, when Fe<sup>2+</sup> and Ti<sup>4+</sup> ions are situated on neighboring structural sites, notable absorption bands  
 355 develop across the visible and near-infrared spectral regions. These Fe<sup>2+</sup>-Ti<sup>4+</sup> pairs exhibit a band center at  
 356 around 580 nm (see Fig. 5) when the electric field vector E is perpendicular to the crystallographic c-axis  
 357 (E⊥c), but a peak at 700 nm is seen when the electric field vector E is parallel to the crystallographic c-axis  
 358 (E||c) (Dubinsky et al., 2020). Although the theory of the energy levels of an individual transition metal ion  
 359 inside a crystal has been extensively explored, the corresponding theory for ion pairs or clusters within a  
 360 crystal remains underdeveloped (Dubinsky et al., 2020).

361 In the present study, the natural unheated geuda sapphire samples were placed in  
 362 atmospheric conditions and subjected to a maximum temperature of 1650 °C for a duration of 10 h.  
 363 According to the examples presented in Fig. 1, samples G03HS and G04HS exhibited a noticeable increase  
 364 in blue coloration, particularly around the area of brown silks and brown color banding/zoning, after  
 365 heating. On the other hand, the initial blue patch (e.g. samples G01HS and G02HS) became paler blue after  
 366 heating, which might be due to the breakage of initial Fe-Ti pairs in those areas. The other groups, which  
 367 have yellowish and/or milky appearances, revealed an increase in blue color after heating (samples G12SF  
 368 and G18LS, Fig. 1). This blue coloration is attributed to two distinct factors, notably the decomposition of  
 369 silk inclusions and a subsequent charge transfer mechanism (Emmett and Douthit, 1993; Hughes, 1997,  
 370 2017; Nassau, 1980, 1981; Themelis, 2018). The process of charge transfer (Ferguson and Fielding, 1972;  
 371 Nassau, 1981) is described as:



373 It is important to note that the blue color observed in sapphire could also be produced  
 374 with the application of heat in oxidizing conditions at high temperature. Heat treatment can be classified as  
 375 high- or low-temperature according to the decomposition of rutile silks in corundum (Nassau, 1981;  
 376 Emmett and Douthit, 1993; Emmett et al., 2003; Hughes, 2017; Hughes and Perkins, 2019). The term low-  
 377 temperature heat treatment has been used (typically referred to as below 1000 °C) when rutile particles still  
 378 reveal their original structures. On the other hand, temperatures beyond 1350 °C denote high-temperature  
 379 heat treatment when rutile silks start to decompose and dissolve within the corundum host (Hughes, 2017;  
 380 Themelis, 2018). Consequently, internal diffusion (indicated by a colored halo surrounding the crystal  
 381 inclusion), molten or altered inclusions, and/or broken silk are strong indicators of high-temperature heat  
 382 treatment. However, low-temperature heat treatment can also produce various altered mineral inclusions  
 383 (Kammerling et al., 1990; McClure and Smith, 2000; McClure et al., 2010; Pisutha-Arnond, 2017;  
 384 Soonthorntantikul et al., 2019).

385 In recent studies, the possibility of employing Fe<sup>2+</sup>-Fe<sup>3+</sup> charge transfer as an alternate  
 386 method for blue coloration has also been mentioned (Nikolskaya et al., 1978; Schmetzer and Kiefert, 1990;  
 387 Häger, 1992, 2001; Sripoonjan et al., 2014; Pisutha-Arnond, 2017). However, it is necessary to emphasize  
 388 that this approach was considered highly improbable (Nassau, 1981). Nevertheless, previous studies have  
 389 indicated that a minor proportion of geuda sapphire from Sri Lanka and geuda-like sapphire from Mogok  
 390 in Burma revealed an alteration in color to blue when subjected to heating in an oxidizing environment  
 391 (Hughes, 1997, 2017; Kyi et al., 1999), which is in complete contradiction to the treatment method  
 392 employed for the geuda sapphire in a reducing condition. The appearance of certain stones displaying a

393 blue coloration under oxidizing conditions might be attributed to the presence of ilmenite silks, which is  
 394 composed of Fe and Ti, with Fe in its reduced  $\text{Fe}^{2+}$  state (Hughes, 1997). Therefore, it is unnecessary to  
 395 reduce  $\text{Fe}^{3+}$  to  $\text{Fe}^{2+}$  ions to generate the  $\text{Fe}^{2+}\text{-Ti}^{4+}$  pairs that are responsible for the manifestation of the blue  
 396 color. Hence, the blue areas have a substantial concentration of Fe ions in form of Fe-Ti pairs, derived from  
 397 the decomposed ilmenite silk inclusions.

398 According to Nassau (1981) and Koivula (1987), the presence of blue dots in heated  
 399 sapphire is attributed to remains of dissolved silk inclusions and internal cation diffusion. The diffusion  
 400 process is positively correlated with temperature and duration of heat treatment (Nassau, 1981). Despite  
 401 the slow diffusion rates of Fe and Ti, the distances across are extremely short, i.e., just a few micrometers  
 402 (Nassau, 1981). Consequently, a potential Fe-Ti combination within the corundum's lattice may generate  
 403 the blue dots.

404 The presence of melt inclusions among the blue dots after high-temperature heating  
 405 might be due to the decomposition of brown silk and its solubility into the host sapphire as demonstrated  
 406 by Jung et al. (2009). They predicted a phase relationship within the  $\text{Al}_2\text{O}_3\text{-Ti}_2\text{O}_3\text{-TiO}_2$  system based on  
 407 experimental data and thermodynamic calculation. Consequently, they suggested that a liquid phase (the  
 408 composition of the liquid inclusion phase varies significantly between  $\text{Al}_2\text{O}_3$  and  $\text{Ti}_3\text{O}_5$ ) could possibly be  
 409 present at a temperature of 1600 °C and slightly below, which is close to the heating temperature (1650 °C)  
 410 of our experiment. Silk inclusions as represented by  $\text{Ti}_2\text{O}_3\text{-TiO}_2$  components may have dissolved into the  
 411 host sapphire ( $\text{Al}_2\text{O}_3$  component), and produced a proper composition of solution which could be melted  
 412 partially at  $\leq 1650$  °C. Some of these melts can be preserved as inclusions after cooling down.

413

### 414 **4.3 Luminescence of sapphire**

415 Luminescence of corundum may be assigned to two types, namely (a) emissions of  
 416 impurity-related centers such as  $\text{Ti}^{4+}$  (commonly known) and (b) emissions of defect-related centers, which  
 417 typically involve either vacancies, such as oxygen (O) or aluminum (Al) vacancies known as F center  
 418 (color center; from the German "*Farbzentrum*"), or interstitials ( $\text{Al}_i$  and  $\text{O}_i$ ), possibly trapped at impurities  
 419 (less known), or both (Vigier et al., 2021a-c). This means that defect-related emission centers in corundum  
 420 refer to an inconsistency in the atomic arrangement limited to one or a few atoms (often called color-  
 421 centers). O vacancies (or electron holes) are sometimes called hole centers, because the holes simply  
 422 designate the absence of an electron. The holes are sometimes filled with one or two electrons in order to  
 423 maintain electroneutrality (Vigier et al., 2021a).

424 As presented in Fig. 8, a notable orange to red luminescence is easily noticeable under  
 425 LWUV excitation in most unheated sapphire samples, except for those of the HS group, which appear inert.  
 426 After heat treatment, the orange to red luminescence that is initially observed in all samples of the LS group  
 427 and many samples of the SF group turns into a purplish-red luminescence. In contrast, no orange to red or  
 428 purplish red luminescence is observed in any sample of the HS group both before and after heating.

429 The origin of orange to red luminescence in sapphire remained controversial, with  
 430 varying ideas among researchers (Vigier et al., 2021a, b, 2023). The occurrence of orange luminescence  
 431 has been documented in some previous studies (e.g. Spencer, 1927; Kane, 1982; Emmett et al., 2003;  
 432 Fritsch et al., 2003; Nasdala and Fritsch, 2024). In the beginning, it was hypothesized that this luminescence

433 is associated with the geographic origin of yellow sapphire from Sri Lanka (Webster, 1984). Subsequently,  
434 Segura (2013) presented an alternative argument to this notion, suggesting the presence of orange  
435 luminescence in various colors of corundum, regardless of treatment or synthetic origin, might be attributed  
436 to the existence of some defects. However, the orange to red luminescence observed in our study  
437 (characterized by a broad emission band) seems to be associated with complex defect-related centers.

438 Orange to red luminescence in sapphire is not due to impurities (Vigier et al., 2021a, b).  
439 HS sapphire (e.g., with ilmenite,  $\text{FeTiO}_3$ ) lack noticeable luminescence, likely because  $\text{Fe}^{2+}$  suppresses  
440 luminescence, contrasting with LS and SF sapphire, which display stronger luminescence both before and  
441 after heating. While sample G23SF shows decreased purplish-red luminescence after heating, most display  
442 increased purplish-red luminescence, potentially due to complex, defect-related centers in the sapphire  
443 lattice. Observations suggested that  $\text{Fe}^{2+}$  acts as a luminescence quencher in the orange to red range  
444 (Andrade et al., 2008; Norrbo et al., 2016; Vigier et al., 2021a, b, c; Vigier and Fritsch, 2022); Orange  
445 luminescence generally appears in colorless, low-Fe areas (Segura, 2013; Notari et al., 2003). However, a  
446 definitive explanation remains unresolved. Regarding blue luminescence, it has been observed that upon  
447 exposure to SWUV light, all natural unheated samples appeared inert. After heating, apart from HS  
448 sapphire, a distinct blue luminescence has been detected throughout most heated sapphire samples (Fig. 8).  
449 Previous studies suggest that luminescence in sapphire becomes noticeable only at heating temperatures of  
450  $1000^\circ\text{C}$  (Hughes and Perkins, 2019), at which point blue luminescence is linked to heat treatment detection.  
451 This luminescence is believed to arise from silk inclusions composed of  $\text{TiO}_2$ , commonly found in natural  
452 blue sapphire. Notably, despite the relatively low Ti concentration (0.02–0.03 wt% oxide) in comparison  
453 to Fe (0.05–0.08 wt% oxide) in some samples (e.g., G12SF and G21LS), blue luminescence remains  
454 detectable. In contrast, HS samples (e.g., G02HS) show an absence of blue luminescence, likely due to the  
455 presence of ilmenite, supporting findings by Norrbo et al. (2016), Andrade et al. (2008), as well as Vigier  
456 et al. (2021a–c; 2023) that  $\text{Fe}^{2+}$  acts as a luminescence quencher. Blue luminescence has been associated  
457 with the interaction between  $\text{O}^{2-}$  and  $\text{Ti}^{4+}$  ions (Evans, 1994; Wong et al., 1995b; Nasdala and Fritsch,  
458 2024), followed by a later hypothesis of a charge transfer process involving  $\text{Ti}^{4+}$  ions and certain defect-  
459 related centers (Lacovara et al., 1985; Mikhailik et al., 2005). However, it was widely accepted that the  
460 blue luminescence (characterized by a broad emission band at blue to green region) observed in sapphire  
461 under SWUV illumination is associated with the presence of Ti impurities, which are classified as element-  
462 related defects (Vigier et al., 2021a, b). Thus, it is likely that the blue luminescence reported in this work  
463 is associated with Ti impurities, whereas orange to red luminescence seems to be associated with complex  
464 defect-related emission centers.

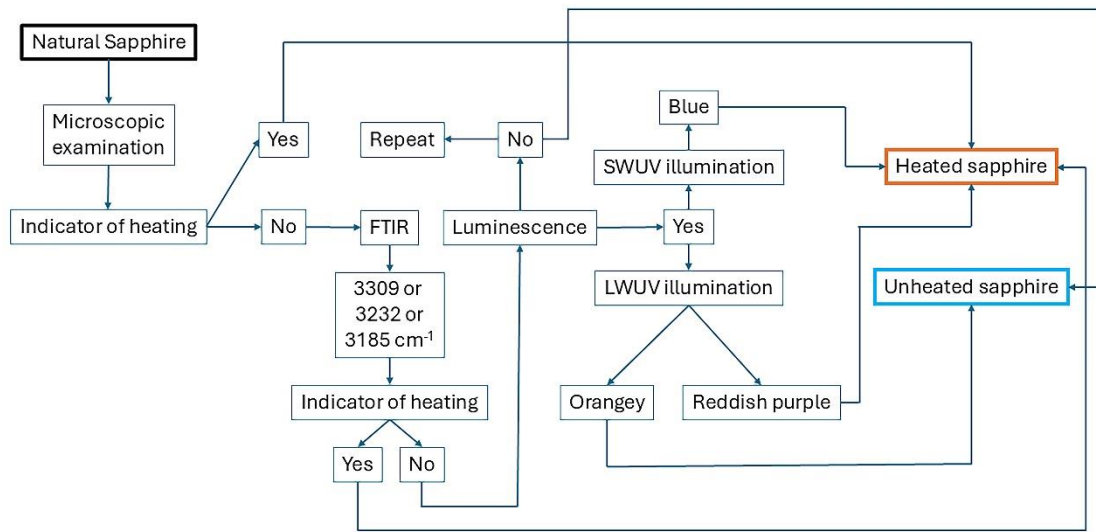
465 The correlation between the orange to red PL emission band (approx. 650 nm, Fig. 9 blue  
466 lines) and orange to red luminescence in unheated sapphire (Fig. 8), as well as the emission band (approx.  
467 525 nm, Fig. 9 red lines) and blue luminescence in heated sapphire (Fig. 8), is particularly evident in the  
468 LS group (Fig. 9b). In contrast, the HS group shows a reduction in emission across the visible spectrum  
469 after heating, indicating inertness under LWUV and SWUV excitation (Fig. 9a). The SF group exhibits a  
470 notable increase in the red emission band (Fig. 9c) and intense purplish-red luminescence under LWUV  
471 excitation after heating. Interestingly, this group also displays strong blue luminescence under SWUV  
472 excitation despite the absence of a corresponding blue emission band, likely due to the 325 nm excitation

473 laser used in our PL investigation. Variations in excitation wavelengths significantly affect observed  
474 emissions, as noted by Wong et al. (1995a) and Vigier et al. (2023), who showed that their sapphire  
475 emission band at 425 nm was only visible with a 254 nm excitation laser. Utilizing distinct SWUV (225  
476 nm) and LWUV (365 nm) lasers, or conducting excitation spectroscopy, may yield more accurate results  
477 compared to relying solely on a 325 nm laser. Thus, the presence of orange to red luminescence at approx.  
478 650 nm and blue luminescence at around 525 nm are vital indicators for differentiating unheated and heated  
479 sapphire.

480

## 481 **5 Conclusions**

482 The present study demonstrates that melt inclusions (~1  $\mu\text{m}$ ) serve as indicators of heat treatment  
483 in sapphires and highlights the critical role of luminescence in distinguishing unheated from heated geuda  
484 sapphire. Under LWUV light, orange luminescence may arise from defect-related F centers, while blue  
485 luminescence under SWUV light likely correlates with Ti impurities. Geuda sapphires with low Fe  
486 concentrations exhibit distinct luminescence, whereas those with HS inclusions show minimal  
487 luminescence due to  $\text{Fe}^{2+}$  quenching effects. The presence of orange luminescence may be a helpful  
488 indicator for unheated geuda sapphires, while blue luminescence is generally absent in unheated samples,  
489 confirming its utility for identifying heated geuda sapphire. Although the  $3309\text{ cm}^{-1}$  O–H stretching band  
490 from FTIR analysis alone is insufficient for differentiation, increased intensity around the 580 nm of an  
491 optical spectra effectively indicates heat treatment, as it corresponds to higher Fe–Ti pair concentrations  
492 from silk inclusion decomposition. Combining blue and/or purplish–red luminescence with additional  
493 analytical techniques provides a promising strategy for accurately distinguishing between unheated and  
494 heated geuda sapphires (Fig. 10). Future research should acquire emission and excitation spectra on the  
495 samples before and after heat treatment. Further investigation of luminescence characteristics from various  
496 sapphire origins (and colors) and clarify the specific Fe and Ti concentrations impacting luminescence is  
497 also recommended. Finally, using as low as 200–254 nm laser excitation may enhance the detection of  
498 emission shifts towards the blue region in heated sapphires, improving gemological identification criteria.



499

500 **Figure 10.** Flowchart proposing the combined strategy for heated sapphire identification criteria.

501

502 *Author Contributions.* T.P., C.S., B.W., L.N. conducted conceptualization, E.G.Z. acquired samples, T.P.,  
 503 C.S., B.W., L.N., C.C.N., M.W., E.L., G.G., T.S. conducted analyses and evaluation, T.P. wrote the  
 504 manuscript, all co-authors reviewed and edited the manuscript.

505

506 *Competing interests.* The corresponding author has declared that none of the authors has any competing  
 507 interests.

508

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519

520 *Conflicts of Interest.* The authors declare that they have no conflict of interest.

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